

# U-type Unileg Thermoelectric Module: A Novel Structure for High-temperature Application with Long Lifespan

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## Abstract:

Strong thermal stress caused by high temperature and difference of thermal expansion coefficient (CTE) will negatively influence the lifespan of thermoelectric modules. In this work, a new high-temperature CaMnO<sub>3</sub>-based U-type unileg thermoelectric module, combining an unileg structure with pn-junction, is proposed and investigated. The novel design avoids the device failure due to different CTEs and high temperature gradients. As a result, the maximal thermal stress ( $\sigma_{\max, \text{TEM}}$ ) of 3.31GPa and fatigue life of 41686 cycles are 46% and 132% of those of traditional modules at 6W and 300K, respectively. To further relieve stress concentration, the effect of rounded corners ( $r_u, r_l$ ), Ag layer thickness ( $H_{\text{Ag}}$ ) and length of right legs ( $L_R$ ), have been studied. It has been found that larger  $r_u$ , and  $r_l$  are suitable to relieve the local stress concentration, and the lowest  $\sigma_{\max, \text{TEM}}$  and highest power ( $P_{\max}$ ) are achieved at  $(r_u, r_l) = (0.1, 0)$  and  $(0, 0.5)$ . Moreover, larger  $L_R$  and  $H_A$  are beneficial for mechanical properties by decreasing the peak stress and dispersing the high thermal stress regions, while module performance is improved at lower  $L_R$  and  $H_{\text{Ag}}$ . Results obtained from this U-type unileg thermoelectric module should influence and guide the design and optimization of high-temperature thermoelectric generators.

**Keywords:** Thermoelectric module; U-type unileg structure; finite-element simulation; thermal stress; lifespan analysis.

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## 1. Introduction

Renewable energy sources, and technology for their exploitation, are urgently needed due to the resources shortage, environmental deterioration, and ecological damage [1-3]. Thermoelectric power generator (TEG), as a novel energy conversion technology, can realize the direct conversion between heat and electricity [4]. In contrast to traditional waste heat recovery technologies, the TEG is advantageous because of its compactness, noiselessness, vibration-less operation, no moving mechanical components, and low ecological hazards. Thus, TEGs have been applied in micropower generators, the internet of things, automobiles, spacecrafts, *etc.*, in last decades [5-7].

The thermoelectric modules, as the main components of TEGs, are typically composed of thermoelectric legs, electrodes and ceramic substrates. Based on the Seebeck effect, thermoelectric modules can directly convert waste heat into useful electricity when a temperature difference ( $\Delta T$ ) is established across the module [8]. The theoretical conversion efficiency ( $\eta$ ) mainly depends on  $\Delta T$  and figure of merit of thermoelectric materials ( $ZT$ ); it is enhanced and approaches the Carnot efficiency as  $\Delta T$  increases; consequently, larger  $\Delta T$  benefits to obtain higher power output. However, higher temperatures will induce larger thermal stresses within the material and among the different materials in the thermoelectric module, resulting in serious deformations, fractures and even device failure. Thus, much new designs and strategies have been proposed to solve the issue of these high stresses.

Poor contact between different materials is a crucial reason for high stress and low stability. Clin *et al.* [9] proved that the stress distribution in the thermoelectric legs was greatly affected by thermal expansion coefficient (CTE) mismatch between electrodes and thermoelectric materials. In this case, Chavez *et al.* [10] proposed a new pn-junction TEG concept. In a pn-junction TEG, as shown in **Figure 1(a)**, the metal contact and substrate on the hot side are diminished, and the electrical connection is made by a direct junction of the p- and n-type thermoelectric materials. This design completely gets rid of contact issues between metal and semiconductor on the hot side and, therefore, shows great potential for constructing reliable and

long-lifetime thermoelectric devices.

Besides, the contact issues of thermoelectric material/electrode interface, and the CTE difference between n- and p-type thermoelectric materials, are also a big challenge for low-stress design. To solve this problem, Marquesn *et al.* proposed an unileg structured module for the first time in 2007. In 2008, Lemonnier *et al.* [11] fabricated and tested a prototype of oxide thermoelectric module only composed of  $\text{Ca}_{0.95}\text{Sm}_{0.05}\text{MnO}$  n-legs. The unileg structure reduces the problems associated with CTE difference between p- and n-type thermoelectric legs. As a result, this architecture contributed to good mechanical strength and increased lifespan during thermal cycling. Afterwards, an improved unileg-TEG design was proposed by Wijesekara *et al.* [12], as shown in **Figure 1(b)**. Here, the integration of electrode and conductor reduced the total number of contacts in the thermoelectric device and further decreased the thermal stress. Therefore, the unileg module with pn-junction can be considered as a promising structure for low-stress design [13].

As mentioned above, the high stress intensity can be effectively reduced through novel structural design. However, for any structure, the stress at the local structure such as the edge of a hole, or at a right-angle beam, has a higher value than the remote stress [14, 15]. The phenomenon of stress concentration will result in fatigue cracks and the failure of an object or component. In this case, Al-Merbati *et al.* [16] proposed that thermal efficiency is improved for certain geometric configuration of the device. Yilbas *et al.* [17] investigated the impact of tapered and rectangular pin configurations on thermal stress in thermoelectric generators, and found that thermal stress developed in tapered configuration attains lower values than that of rectangular cross-section. Wang *et al.* [18] focused on studying the feasibility of an X-type thermoelectric module with different draft angles. They found that the X-type structure can enhance the performance of the thermoelectric module with regard to both electrical power and mechanical reliability. The effect of soldering thickness has been researched by Wu *et al.* [19], and they proposed choosing a suitable tin soldering thickness will not only alleviate thermal stress intensity in the module, but also increases thermal efficiency. All in all, unreasonable geometric structure is a main

reason for high local stress, which will degrade the module performance and lifespan [20-22]. Geometry optimization, such as rounded or chamfered corners and conductor with optimal thickness, is a good method to solve problems of stress concentration.

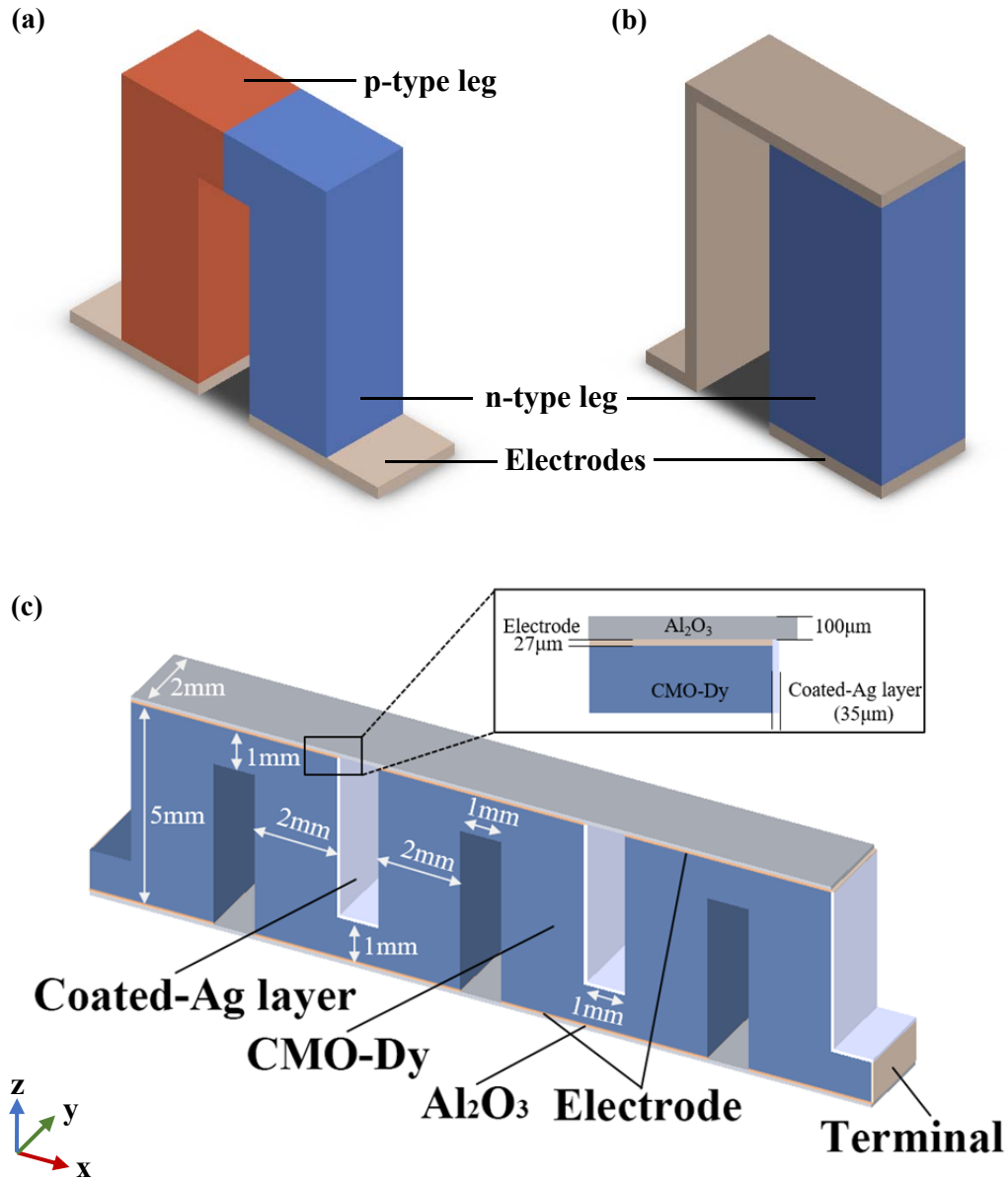
In this work, an U-type unileg thermoelectric generator, combining an unileg structure with pn-junction, is proposed and investigated to relieve the crucial problem of high thermal stress and short lifespan caused by CTE difference or high heat flux. Moreover, the geometric parameters, such as rounded corners ( $r_u, r_l$ ), thickness of Ag layer ( $H_{Ag}$ ), and length of right legs ( $L_R$ ), are optimized to further relieve stress concentration. The results showed that the U-type unileg thermoelectric module has excellent mechanical strength and long lifespan, which would be highly beneficial for commercializing high-temperature thermoelectric devices.

## 2. Simulation approach

### 2.1 Numerical model

The schematic diagram of the U-type unileg thermoelectric module with  $r_u=r_l=0\text{mm}$ , as an example, is constructed and drawn by COMSOL and SOLIDWORKS in **Figure 1(c)**. The designed thermoelectric module is assembled with  $\text{Al}_2\text{O}_3$  plates, Ag electrodes, and Dy-doped  $\text{CaMnO}_3$  unileg with an Ag-coated side. The dimensions ( $l_x \times l_x \times l_z$ ) of upper and lower  $\text{Al}_2\text{O}_3$  plates are  $17 \times 2 \times 0.1\text{mm}^3$  and  $19 \times 2 \times 0.1\text{mm}^3$ , respectively. The electrodes, with  $5 \times 2 \times 0.027\text{mm}^3$  size are used to connect the unilegs and  $\text{Al}_2\text{O}_3$  plates, while the terminal electrodes with  $0.027 \times 2 \times 1\text{mm}^3$  size, are applied to connect load resistance. This Ag-coated unileg design is the main part of the U-type module studied in this work. Here, three interconnected thermoelectric unilegs are obtained from a single block of material. The spacing between adjacent thermoelectric materials are alternatively connected to the upper or lower substrate, having  $1 \times 2 \times 4.027\text{mm}^3$  dimensions. The Ag-coating layer, with  $35\mu\text{m}$  thickness, is deposited on the right side of each unileg, and the junction region between two unilegs, which is used for electric conduction. In other words, three left thermoelectric legs are connected through Ag-coated right legs electrically in series. The thickness of electrode and Ag-coating layer are chosen from previously reported experimental articles [7, 23]. The

temperature-dependent thermoelectric properties of Dy-doped  $\text{CaMnO}_3$  are shown in **Figure S1** [24]. The density ( $\rho_d$ ), Young's modulus ( $E$ ), Poisson's rate ( $\nu$ ), and CTE, of Dy-doped  $\text{CaMnO}_3$  (CMO-Dy), Ag and  $\text{Al}_2\text{O}_3$  are presented in **Table S1**.



**Figure 1** The schematics of (a) a  $\pi$ -type thermoelectric module with pn-junction; (b) a unileg thermoelectric module; and (c) a U-type unileg thermoelectric module with straight corners.

## 2.2 Finite-element simulation

Establishing the relationship between module design and physical properties is the main objective in the simulation. Some boundary conditions and reasonable assumptions are adopted for this purpose. Firstly, the thermoelectric module is in

steady state. Secondly, the thermoelectric module is regarded as an adiabatic system where the heat radiation and convection on all surfaces, except on the hot and cold ends, are ignored. Thirdly, the upper and lower sides of module are considered as the hot and cold ends. The absorbed heat in the hot side ( $Q_h$ ) is assumed to be constant, 6W, and the cold-end temperature is fixed at 300K. Fourthly, the electrical and thermal contacts between thermoelectric materials and Ag-coating layer, or Ag electrodes, are considered [25]. The surface roughness and surface roughness slope are, respectively, 0.08 $\mu$ m and 0.544, according to previously reported experimental results [26]. Fifthly, the leftmost electrode is grounded, and rightmost electrode is terminal. Sixthly, the bottom of the U-type unileg module is fixed.

The heat flux and electric current coupled phenomena in the thermoelectric module can be well calculated and analyzed through the governing equations shown below [27, 28].

$$\nabla \cdot \mathbf{q} = \mathbf{J}^2 / \sigma \quad (1)$$

$$\nabla \cdot \mathbf{J} = 0 \quad (2)$$

In here,  $\mathbf{J}^2 / \sigma$  is the term of Joule heat. The heat flux vector ( $\mathbf{q}$ ) and current density vector ( $\mathbf{J}$ ) in three dimensions can be calculated as follows.

$$\mathbf{q} = -\kappa(T) \nabla T + P_c(T) \mathbf{J} \quad (3)$$

$$\mathbf{J} = -\nabla V / \rho(T) + S \nabla T / \rho(T) \quad (4)$$

where  $S(T)$ ,  $\rho(T)$ ,  $\kappa(T)$  and  $P_c(T)$  are the temperature-dependent Seebeck coefficient, electrical resistivity, thermal conductivity, and Peltier coefficient calculated by  $S(T)T$ , respectively.  $\nabla T$  and  $\nabla V$  are the temperature and electric potential gradients. Based on the above equations, the thermoelectric constitutive equations can be described by Equations (5) and (6),

$$\nabla \cdot (\kappa(T) \nabla T) + \rho(T) \mathbf{J}^2 - T \mathbf{J} \cdot (\partial S / \partial T) \nabla T + \nabla S = 0 \quad (5)$$

$$\nabla \cdot (-\nabla V + S(T) \nabla T) / \rho(T) = 0 \quad (6)$$

Based on simulative results, the module performance results include internal resistance ( $R_{in}$ ), output voltage ( $V_o$ ), working current ( $I$ ), output power ( $P$ ), and maximum output power ( $P_{max}$ ), which are calculated through the following expressions.

$$R_{in} = \int \rho_d(T) dl / dS, \quad T = f(x, y, z) \quad (7)$$

$$V_o = (R_L / (R_{in} + R_{load})) \int S(T) dT \quad (8)$$

$$I = V_o / R_L = \int S(T) dT / (R_{in} + R_{load}) \quad (9)$$

$$P = I^2 R_L = (\int S(T) dT)^2 / (R_{in} + R_{load}) \quad (10)$$

$$P_{max} = (\int S(T) dT)^2 / 4R_{in} \quad (11)$$

Here,  $R_{in}$  depends on the temperature field and current direction;  $dl$  and  $dS$  are unit length along the current direction, and unit cross-sectional area perpendicular to the current direction, respectively;  $R_L$  is the load resistance.

Thermal stress is generated due to the differential thermal expansion of materials forming the thermoelectric generator. The equations governing the displacement-strain relations for the thermal stress can be expressed as shown below [16, 29].

$$\bar{\epsilon}_{xx} = \frac{\partial \bar{u}}{\partial \bar{x}}, \quad \bar{\epsilon}_{yy} = \frac{\partial \bar{v}}{\partial \bar{y}}, \quad \bar{\epsilon}_{zz} = \frac{\partial \bar{w}}{\partial \bar{z}} \quad (12)$$

$$\bar{\epsilon}_{xy} = 0.5 \left( \frac{\partial \bar{u}}{\partial \bar{y}} + \frac{\partial \bar{v}}{\partial \bar{x}} \right), \quad \bar{\epsilon}_{yz} = 0.5 \left( \frac{\partial \bar{w}}{\partial \bar{y}} + \frac{\partial \bar{v}}{\partial \bar{z}} \right), \quad \bar{\epsilon}_{zx} = 0.5 \left( \frac{\partial \bar{w}}{\partial \bar{x}} + \frac{\partial \bar{u}}{\partial \bar{z}} \right) \quad (13)$$

The stress-strain relation can be expressed in a dimensionless form using a nonsymmetrical Jacobian matrix

$$\begin{Bmatrix} \bar{\sigma}_{xx} \\ \bar{\sigma}_{yy} \\ \bar{\sigma}_{zz} \\ \bar{\sigma}_{yz} \\ \bar{\sigma}_{zx} \\ \bar{\sigma}_{xy} \end{Bmatrix} = \frac{\bar{E}}{(1+\nu)(1-2\nu)} \begin{bmatrix} 1-\nu & \nu & \nu & 0 & 0 & 0 \\ \nu & 1-\nu & \nu & 0 & 0 & 0 \\ \nu & \nu & 1-\nu & 0 & 0 & 0 \\ 0 & 0 & 0 & 1-2\nu & 0 & 0 \\ 0 & 0 & 0 & 0 & 1-2\nu & 0 \\ 0 & 0 & 0 & 0 & 0 & 1-2\nu \end{bmatrix} \times \begin{Bmatrix} \bar{\epsilon}_{xx} \\ \bar{\epsilon}_{yy} \\ \bar{\epsilon}_{zz} \\ \bar{\epsilon}_{yz} \\ \bar{\epsilon}_{zx} \\ \bar{\epsilon}_{xy} \end{Bmatrix} - \begin{Bmatrix} 1 \\ 1 \\ 1 \\ 0 \\ 0 \\ 0 \end{Bmatrix} \frac{\alpha \bar{E} \bar{T}}{1-2\nu} \quad (14)$$

The three principal stresses are represented as  $\sigma_1$ ,  $\sigma_2$ , and  $\sigma_3$ , respectively. The von Mises equivalent stress can be obtained from the fourth strength theory of mechanics of materials also known as the distortion of energy theory. It describes the total combined stresses in all three dimensions,

$$\sigma = \sqrt{\frac{[(\sigma_1 - \sigma_2)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2]}{2}} \quad (15)$$

The fatigue limit can be provided by a combined Basquin and Coffin-Manson relation

[30-32], given by

$$\varepsilon_a = \frac{\sigma'_f}{E} (2N_f)^b + \varepsilon'_f (2N_f)^c \quad (16)$$

where  $\sigma'_f$  is the fatigue strength coefficient,  $b$  is the fatigue strength exponent,  $\varepsilon'_f$  is the fatigue ductility coefficient,  $c$  is the fatigue ductility exponent, and  $E$  is the Young's modulus.  $N_f$  is the number of load reversals, and thus  $2N_f$  is the number of full cycles to failure,  $N$ , at a strain amplitude of  $\varepsilon_a$ . At low strains the fatigue life is limited by a Cycle Cutoff. It is well known that the mean stress has a significant effect on the fatigue life. Morrow proposed a mean stress correction to the Basquin part of the combined Basquin and Coffin-Manson relation according to

$$\varepsilon_a = \frac{\sigma'_f - \sigma_{ave}}{E} (2N_f)^b + \varepsilon'_f (2N_f)^c \quad (17)$$

where  $\sigma_{ave}$  is the mean stress of the load cycle.

### 2.3 Model validation

To verify the correctness of the simulation method, an experimental  $\text{GdCo}_{0.95}\text{Ni}_{0.05}\text{O}_{3-\delta}/\text{CaMn}_{0.98}\text{Nb}_{0.02}\text{O}_{3-\delta}$  thermoelectric module is simulated and analyzed by the above same mathematical model at  $\Delta T$  of 500K [33]. In addition, the distribution of von Mises stress for a  $\pi$ -type thermoelectric module studied by Karri *et al.* is also simulated and analyzed at  $T_h=250^\circ\text{C}$  and  $T_c=100^\circ\text{C}$  [34]. When comparing these results to the simulated ones, presented in **Figures S2** and **S3**, the distributions of  $R_{in}$ , open-circuit voltage ( $V_{oc}$ ),  $P_{max}$  and maximum von Mises stress ( $\sigma_{max}$ ), are well matched with those obtained in reference. The deviations between experimental and simulated values on  $R_{in}$ ,  $V_{oc}$ ,  $P_{max}$  and  $\sigma_{max}$  are only 1.62%, 2.86%, 7.42% and 8%, respectively. The deviation is owing to the difference of some initial setting values, which are not described in the reference. Anyway, the main factors and effects have been considered in this simulation work, so the simulated results obtained in this work are credible.



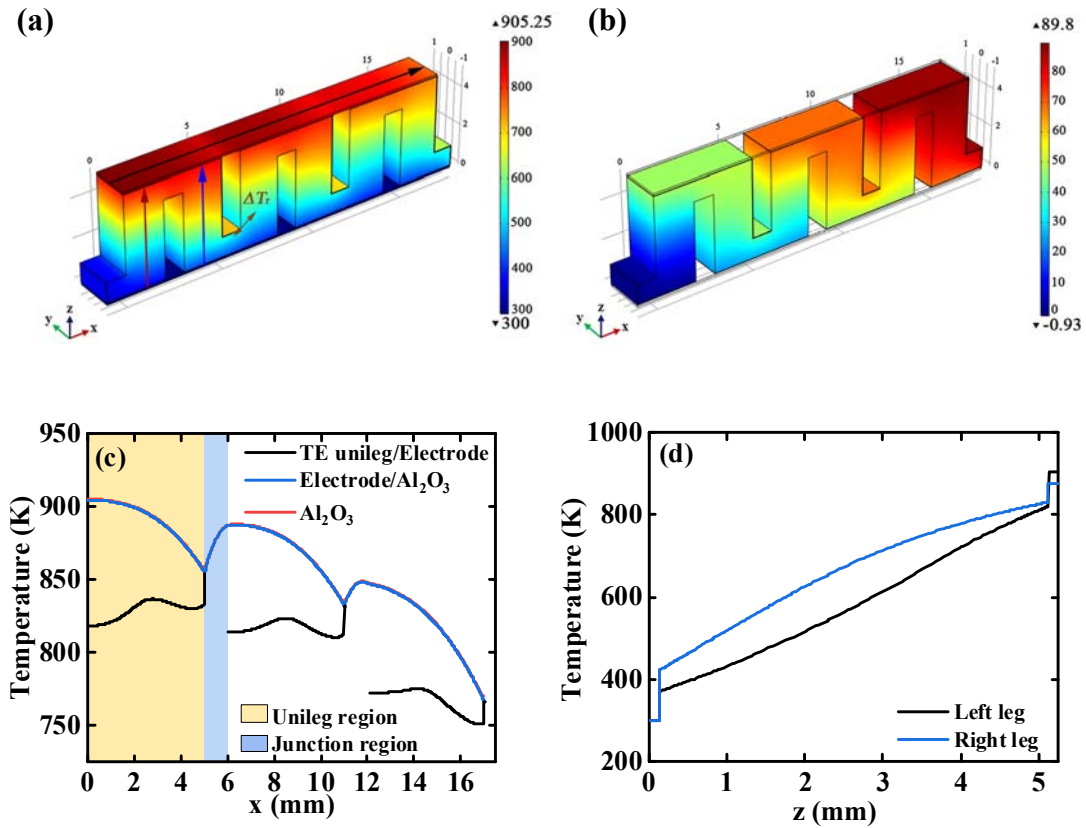
## 2.4 Mesh convergence

Mesh convergence test has been performed to raise confidence in the simulative results of the U-type unileg module. As a result, the average hot-side temperature ( $T_{h-ave}$ ) and  $P_{max}$  of four meshes consisting of 12800, 42465, 85980, and 108005 cells, are 859.74, 859.76, 859.80, and 859.80K; and 6.613, 6.614, 6.616, and 6.616mW, respectively. Thus, the model with 85980 cells exhibits the best balance in terms of accuracy and computational time, and has been selected for the following simulation.

## 3. The performance of the U-type unileg module

**Figure 2** shows the temperature and voltage distribution of the U-type unileg module without rounded corners at  $Q_h=6W$  and  $T_c=300K$  under open-circuit conditions. As presented in **Figure 2(a)**, the temperature fluctuates up and down along x-axis, while maintained unchanged along y-axis, and monotonously increased along z-axis. It means that the heat flux flow from the hot-side substrate to the left legs and Ag-coated right legs, and then to the lower electrodes and cold-side substrate. To gather more detailed information about the hot-side temperature distribution, the evolution of temperature for the thermoelectric unileg/electrode interface ( $T_{TEU/E}$ ), electrode/ $Al_2O_3$  interface ( $T_{E/A}$ ), and upper side of  $Al_2O_3$  plate ( $T_h$ ) along x-axis (black line in **Figure 2(a)**), are exhibited in **Figure 2(c)**. The  $T_{E/A}$ , and  $T_h$  curves are well coincident due to the high thermal conductivity of  $Al_2O_3$  plate. The temperature drops in the unileg region ( $x=0-5mm$ ,  $6-11mm$ ,  $12-17mm$ ), and then rapidly increases in the junction region between different unilegs ( $x=5-6mm$ ,  $11-12mm$ ). The abrupt change of temperature is caused by the high thermal conductivity of Ag-coating layer and large concentrated heat flux in the junction region. Consequently, the maximum hot-side temperature ( $T_{h,max}$ ) of 905.25K and minimum value ( $T_{h,min}$ ) of 769.94K are obtained at leftmost and rightmost sides of the upper  $Al_2O_3$  plate, respectively, leading to 861.15K average value ( $T_{h,ave}$ ). The variation tendency of  $T_{TEU/E}$  for the three unilegs are similar. The temperature, as actual hot-side temperature of thermoelectric material, in the unileg region is firstly raised, followed by dropping along the x-axis direction, and then showing a sharp increase at the Ag-coating layer position. The peak value

appears at the middle of the unileg region because of the high concentrated heat flux. In addition, the temperature variation curves of left and right legs, for leftmost unileg, along z-axis are presented in **Figure 2(d)**, where the bottom side of the module is set as  $z=0\text{mm}$ . In this figure, the temperature of both legs goes up as height increases. However, the actual temperature differences ( $\Delta T_a$ ) of left and right legs are 450.98, and 410.15K, respectively, which are much lower than the  $\Delta T$  across the module, 561.15K, ( $\Delta T=T_{h,ave}-300\text{K}$ ).

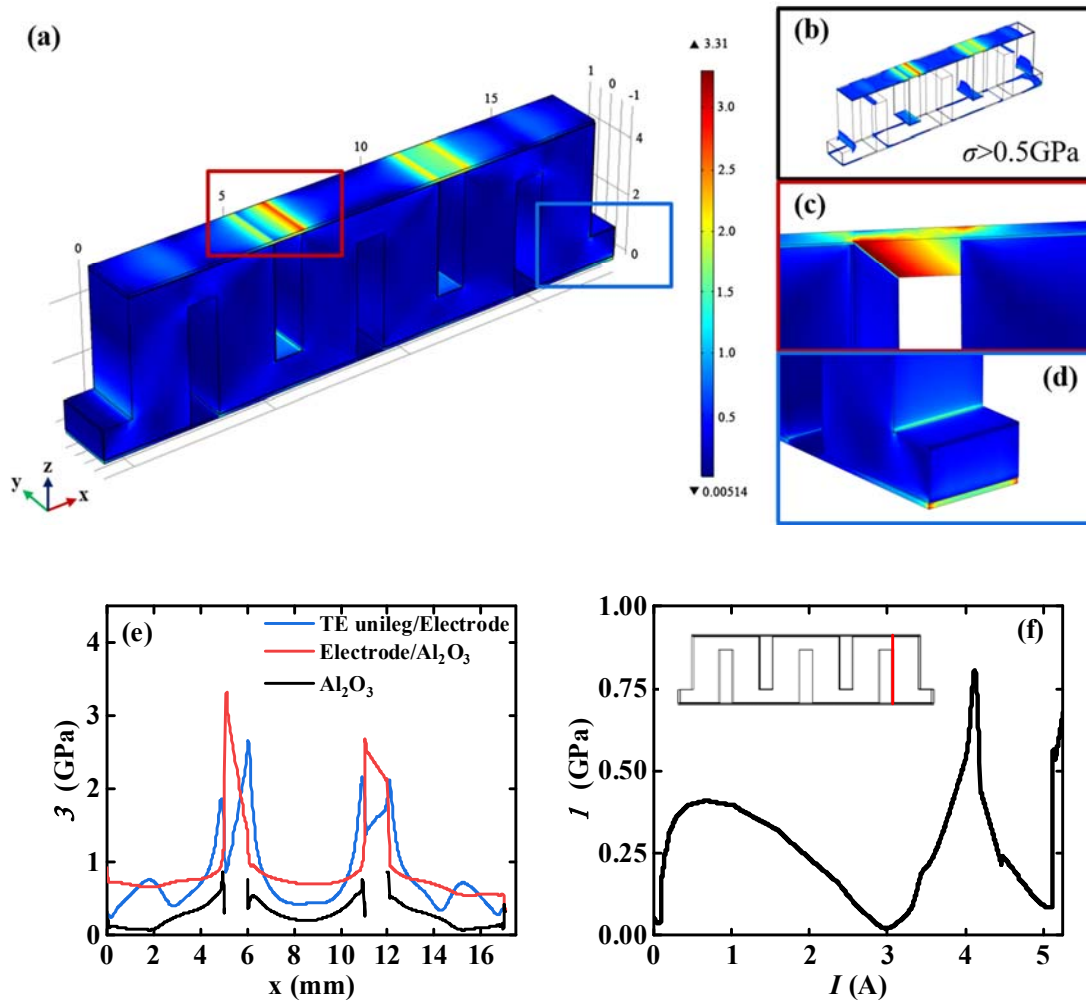


**Figure 2** The (a) temperature; and (b) voltage distributions of for the U-type unileg module under open-circuit conditions. (c) Temperature variation of thermoelectric unileg/electrode interface, electrode/Al<sub>2</sub>O<sub>3</sub> interface, and Al<sub>2</sub>O<sub>3</sub> surface along x-axis. (d) Temperature variation of left and right legs for leftmost unileg along z-axis.

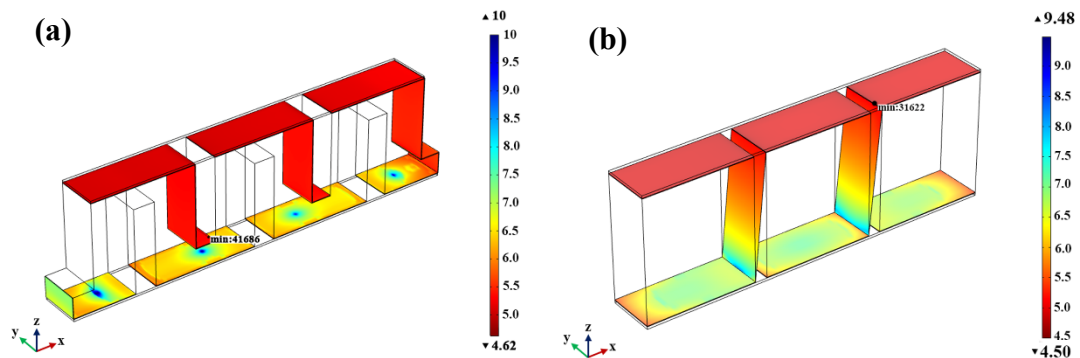
These lower  $\Delta T_a$  values are mainly due to the high thermal contact resistance between electrode and thermoelectric materials. Moreover, the larger  $\Delta T_a$  of left leg compared to the right one is owing to the relatively lower thermal resistance ( $R_{th}$ ) of right leg with Ag-coating layer. Based on the temperature distribution above mentioned, the voltage distribution of the U-type unileg module in open-circuit condition is described

in **Figure 2(b)**. In this figure, the right legs are practically short-circuited and act as electrical conductors in the unileg structure; in other words, the left legs of three unilegs are electrically connected in series through right legs with Ag-coating layer. Consequently, the module will keep working even if the electrode and  $\text{Al}_2\text{O}_3$  in the hot side are broken. Furthermore, the  $V_{oc}$  does not monotonically increase because the reverse temperature difference ( $\Delta T_r$ ) between junction region and the left leg of the adjacent unileg results in voltage losses. Finally, the  $V_{oc}$  generated by these three unilegs is 89.8mV at  $Q_h=6\text{W}$ , with  $T_{h,ave}=861.15\text{K}$  and  $T_c=300\text{K}$ .

**Figure 3(a)** plots the distribution of von Mises stress when the bottom of the module is fixed. Obviously, the stress distribution in the unileg module is inhomogeneous due to the complex geometric structure and temperature field. However, most area is subjected at low  $\sigma$  ( $<0.5\text{GPa}$ ), as shown in **Figure 3(b)**, while the high stresses are locally presented at the hot-side ceramic substrate (**Figure 3(c)**), and the right-angle corners (**Figure 3(d)**). More specifically, the variation of  $\sigma$  for the unileg/electrode interface ( $\sigma_{TEU/E}$ ), electrode/ $\text{Al}_2\text{O}_3$  interface ( $\sigma_{E/A}$ ) and  $\text{Al}_2\text{O}_3$  surface ( $\sigma_A$ ) along x-axis are shown in **Figure 3(e)**. The maximum stress (3.31GPa) is obtained at the Ag-coating layer position ( $x=5\text{mm}$ ), in the electrode/ $\text{Al}_2\text{O}_3$  interface, which is the most likely place for a crack. Considering the free expansion along y-axis and the rigid joints along x- and z-axis, the crack will develop along y-axis.



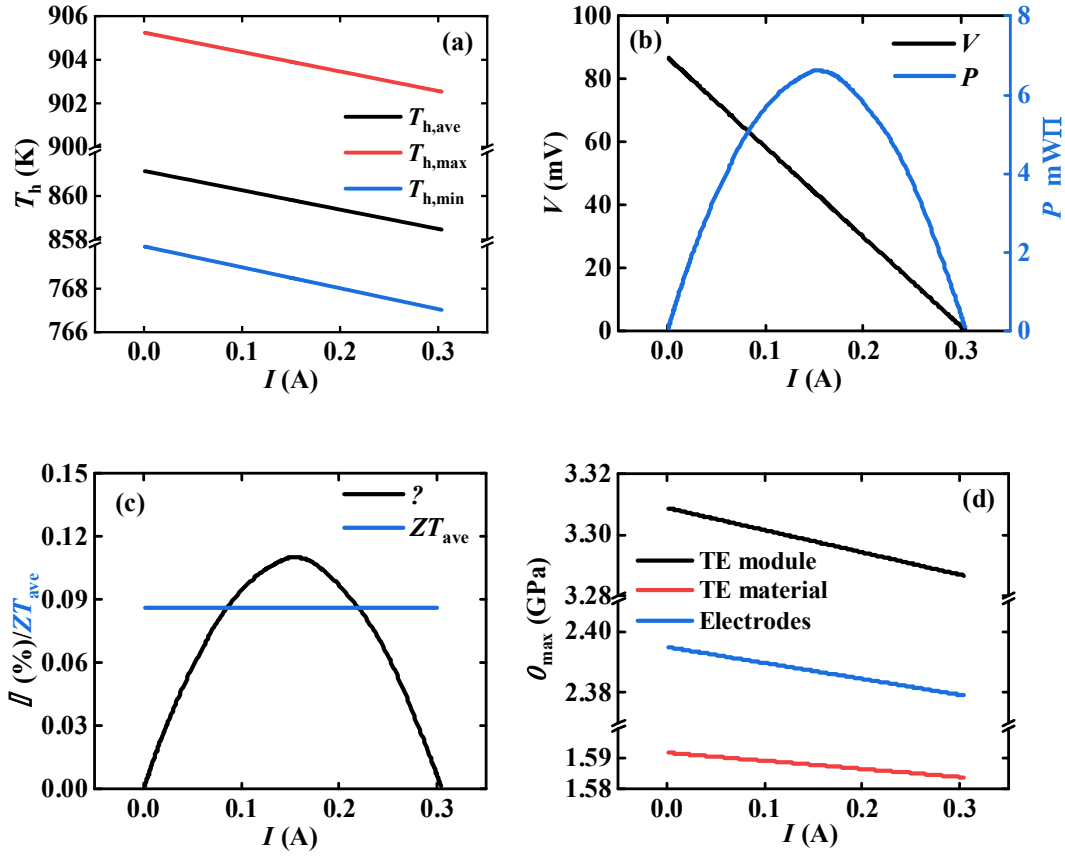
**Figure 3** (a-d) von Mises stress distribution of the U-type unileg module under open-circuit conditions. (e) Variation of von Mises stresses for the unileg/electrode interface, electrode/ $\text{Al}_2\text{O}_3$  interface and  $\text{Al}_2\text{O}_3$  surface along x-axis. (f) Variation of von Mises stress along z-axis.



**Figure 4** The fatigue life of (a) U-type unileg module; and (b) traditional unileg module.

For  $\sigma_{\text{TEU/E}}$  and  $\sigma_{\text{A}}$ , the highest stress values are obtained at  $x=5, 6, 11,$  and  $12\text{mm}$ . The phenomenon of stress concentration is caused by the coupled effect of large temperature gradients, right-angle structure, and CTE difference between Ag electrodes and  $\text{Al}_2\text{O}_3$  substrate. In addition, the variation of  $\sigma$  along  $z$ -axis (red line in the illustration) is presented in **Figure 3(f)**, when the bottom side of the module is  $z=0\text{mm}$ . The  $\sigma$  rises and falls repeatedly as increasing height. The first peak value,  $0.41\text{GPa}$ , occurs at  $z=0.6\text{mm}$  due to the strong constraint caused by the surrounding material; the second peak value,  $0.95\text{GPa}$  as the maximum value, is obtained at  $z=4.1\text{mm}$ , because of the right-angle structure; and the third peak value,  $0.67\text{GPa}$ , emerges on the upper side of the module owing to the high temperature and CTE difference. Compared to the traditional unileg module, as presented in **Figure S4**, the  $\sigma_{\text{max}}$  for both modules are obtained at junction regions of hot end. The  $\sigma_{\text{max}}$  of the overall thermoelectric module ( $\sigma_{\text{max,TEM}}$ ), TE materials ( $\sigma_{\text{max,TE}}$ ) and electrode ( $\sigma_{\text{max,E}}$ ) for traditional unileg module are  $7.15, 1.65$  and  $4.26\text{GPa}$ , which are  $116\%, 3.8\%$  and  $78.2\%$  higher than the obtained for the U-type unileg module, respectively.

**Figure 4** presents the fatigue life of the electrode and Ag-coating layer for U-type unileg module and traditional unileg module to better reflect the module stability. The fatigue life is calculated by formulas 16 and 17 based on the combined Basquin and Coffin-Manson model. Here,  $\sigma'_f, \epsilon'_f, b,$  and  $c$  of Ag are set as  $44.5\text{MPa}, 3.47, -0.1304,$  and  $-0.51$ , according to the previously reported data [30, 31], respectively. In these figures, the smallest cycle number ( $N$ ) of the U-type module is about  $41686$ , which is  $32\%$  larger than that of traditional unileg module. Moreover, due to the special structure of U-type unileg module, the damage of local position will not cause device failure. As for traditional unileg module, the module will stop working when the electrode is damaged. All in all, the low stress and high cycle lifetime show the high stability and commercial value of the U-type unileg thermoelectric modules.



**Figure 5** (a) average, maximum, and minimum hot-side temperature; (b) output voltage, and output power; (c) conversion efficiency, and average figure of merit; and (d) maximum von Mises stress for the U-type unileg thermoelectric module, thermoelectric material, and electrodes, as a function of current.

**Table 1** Materials, couples number, temperature difference, output power and output power density for reported oxide-based modules.

Ref	p/n-type leg	$N$	$\Delta T(K)$	$P_{\max}(mW)$	$p_{\max}(mW/cm^2)$
Lemonnier et al.	$Ca_{0.95}Sm_{0.05}MnO_3$	2	360	16	3.53
Conze et al.	$TiO_x$	20	500	120	13.3
Sharma et al.	$Ca_{0.92}La_{0.08}MnO_3$	9	480	50	12.5
This work	$Ca_{0.9}Dy_{0.1}MnO_3$	3	560	6.6	17.37

Thermoelectric modules as power generators are always used to supply electricity to applications in the loop, thus their output and mechanical performance are studied in a closed circuit. **Figure 5** presents the evolution of  $T_h$ ,  $V$ ,  $P$ ,  $\eta$  and  $\sigma_{\max}$  as  $I$  increases under closed-circuit conditions, when  $Q_h$ , and  $T_c$  are 6W, and 300K, respectively. In

**Figure 5(a)**, it can be observed that  $T_{h,ave}$ ,  $T_{h,max}$ , and  $T_{h,min}$  decreased by about 3K with the increasing of  $I$ . This evolution is due to the coupled impact of Peltier effect and Joule heating. Based on the monotonically decreased temperature,  $V$  is linearly reduced as  $I$  enhances. The  $V$ - $I$  linear curve follows the ohm's law, which proves the correctness of these simulative results. Then,  $P$  is firstly enhanced as  $I$  enlarges and then reverses at  $I=0.15A$ . The  $P_{max}$ , 6.6mW, is obtained at load matching conditions. Compared to previous works about oxide-based unileg modules, as shown in **Table 1**, the U-type thermoelectric module shows a relatively higher electrical performance ( $P_{max}=17.37mW/cm^2$ ). As for  $\eta$ , it firstly increases and then decreases with the increasing of  $I$ . The maximum value is only 0.11%, which is far from meeting the industrial requirements. The low  $\eta_{max}$  is mainly due to two reasons: low  $ZT_{ave}$  of n-type  $CaMnO_3$ -based thermoelectric material (only 0.086), and much unnecessary energy losses caused by Ag-coated right legs and contact resistances. **Figure 5(d)** shows that  $\sigma_{max,TEM}$  drops from 3.31 to 3.29GPa when  $I$  varies from 0 to 0.3A, while  $\sigma_{max,TE}$  is reduced from 1.59 to 1.58GPa. This stress reduction can be explained by the observed decrease of temperature.

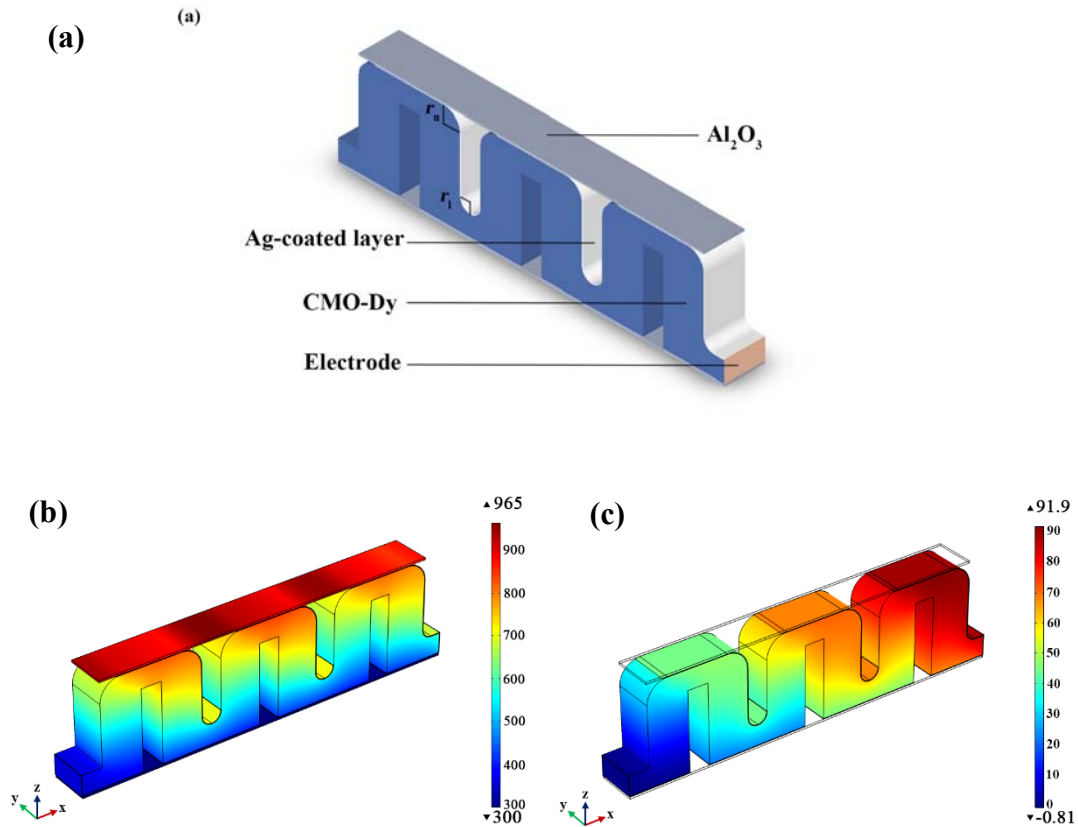
In short, in the U-type unileg structure, one of the legs is practically short-circuited by the Ag-coating layer and acts as an electrical conductor in the structure, avoiding the damage of thermoelectric modules owing to the CTE difference between different components. Moreover, the structure eliminates the need for electrodes at the hot end of the thermoelectric legs, leading to higher stability and promising commercial value.

#### 4. The effect of the radius of round corners

As observed in the previous results, the novel U-type unileg structure is beneficial for relieving stress intensity and enhancing mechanical properties; however, it is found that sharp change in local structural configuration also results in heavy stress concentration. Thus, in order to minimize stress and enlarge lifespan, round chamfering has been considered, and presented in **Figure 6(a)**. In this figure,  $r_u$ , and  $r_l$  are the fillet radius of upper and lower corners, respectively. **Figures 6(b)** and **6(c)** display the temperature and voltage distributions of the U-type unileg module with

$r_u=1\text{mm}$  and  $r_l=0.5\text{mm}$ , under open-circuit conditions, at  $Q_h=6\text{W}$  and  $T_c=300\text{K}$ . Compared to the initial structure, the module has similar temperature and voltage distributions. However, due to the enlarged  $R_{th}$ , the improved U-type module with round chamfering has larger  $T_{h,max}$  (965K) and, consequently, a slightly higher open-circuit voltage (91.9mV).

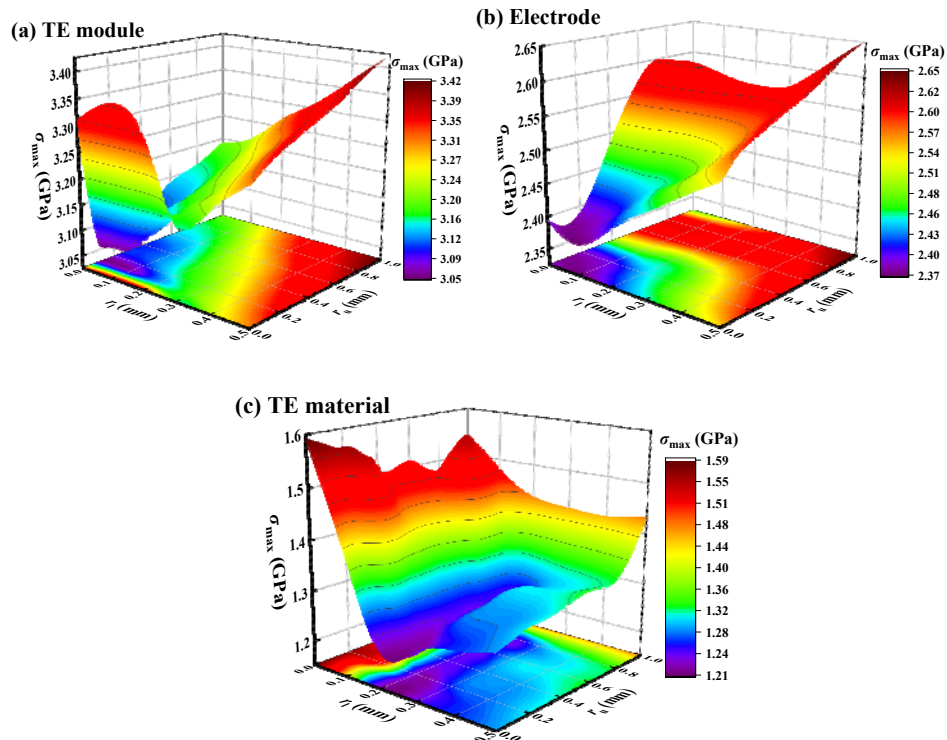
**Figure 7** exhibits the  $\sigma_{max,TEM}$ ,  $\sigma_{max,TE}$ , and  $\sigma_{max,E}$  for the U-type unileg thermoelectric module, as a function of  $r_l$ , and  $r_u$ , at  $Q_h=6\text{W}$  and  $T_c=300\text{K}$ , under load matching conditions. As it can be seen in **Figure 7(a)**, with the enlarging of  $r_u$ ,  $\sigma_{max,TEM}$  firstly drops and then rises at  $r_l=0-0.3\text{mm}$ ; the lowest point is obtained at  $r_u=0.1\text{mm}$ . At  $r_l=0.3-0.5\text{mm}$ ,  $\sigma_{max,TEM}$  shows a monotonously increasing tendency as  $r_u$  enlarges. With the increasing of  $r_l$ , the  $\sigma_{max,TEM}$  is reduced and then enhanced at  $r_u=0\text{mm}$  and, additionally, the value keeps growing for  $r_u>0\text{mm}$ . The minimum value, 3.07GPa, is obtained at  $(r_u,r_l)=(0.1,0)$ .



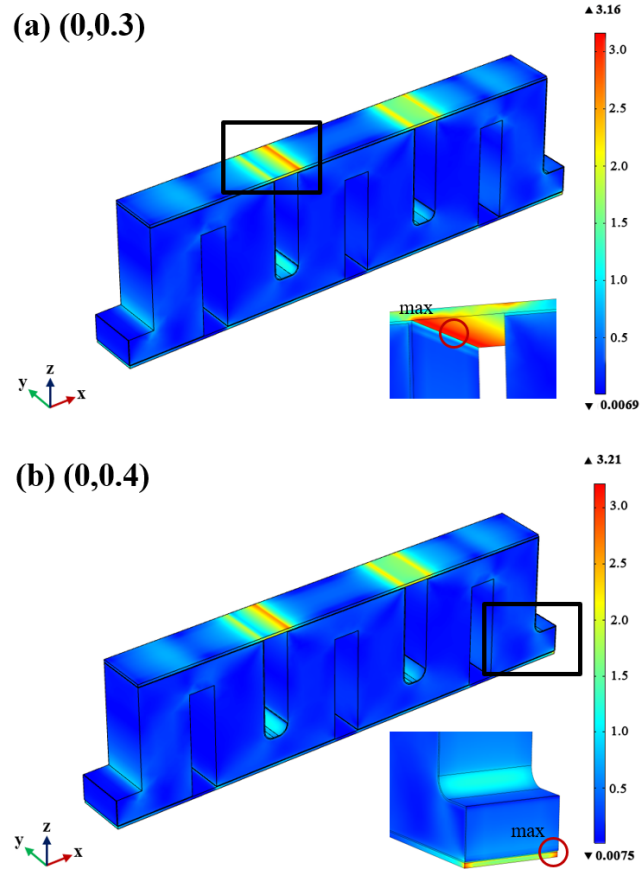
**Figure 6** (a) Schematics of a novel U-type unileg thermoelectric module with rounded corners;(b) temperature; and (c) voltage distribution in the U-type unileg module under open-circuit conditions, when  $Q_h=6\text{W}$  and  $T_c=300\text{K}$ .



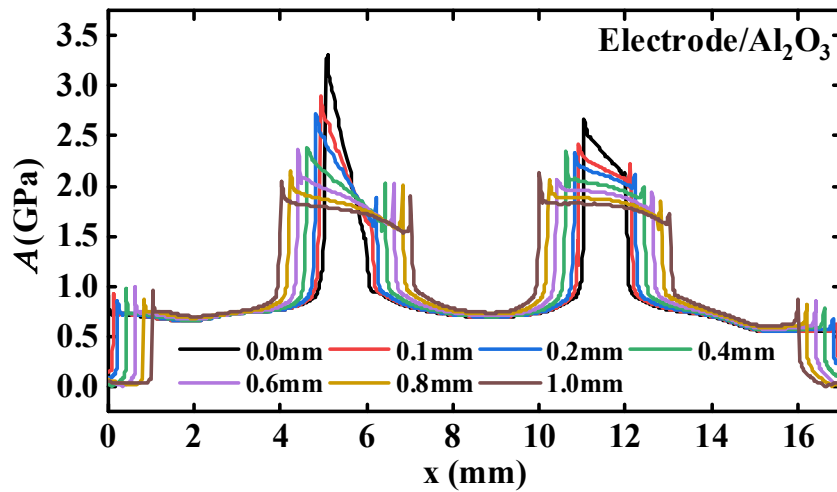
The evolution of  $\sigma_{\max,E}$ , and  $\sigma_{\max,TE}$ , is similar to the observed for  $\sigma_{\max,TEM}$ , reaching the lowest values at (0,0.1), and (0,0.2), respectively. This abnormal evolution is due to fact that  $\sigma_{\max,TEM}$  is obtained at the upper corner position, with low  $r_u$  and  $r_l$ , as presented in **Figure 8(a)**. When increasing  $r_u$  and  $r_l$ , the  $\sigma_{\max,TEM}$  position is shifted to the bottom corner, as described in **Figure 8(b)**. In fact, the larger radius will result in lower concentrated stresses. For the upper corner ( $x=5\text{mm}$ ), as shown in **Figure 9**, the local stress is rapidly and monotonously reduced from 3.31 to 2.05GPa, when  $r_u$  varies from 0 to 1mm. For the lower corner in **Figure 10**, the larger  $r_l$  alleviates the peak thermal stress intensity and shifts the high thermal stress region from the side to the center of the modules. Summarizing, the local stresses of upper and lower corners are effectively decreased by 38% and 45%, respectively, through adding rounded corners. However, for the whole U-type  $\text{CaMnO}_3$  unileg module, the lowest stress (3.07GPa) is achieved at relatively low  $r_u$  (0.1mm), and  $r_l$  (0mm) values.



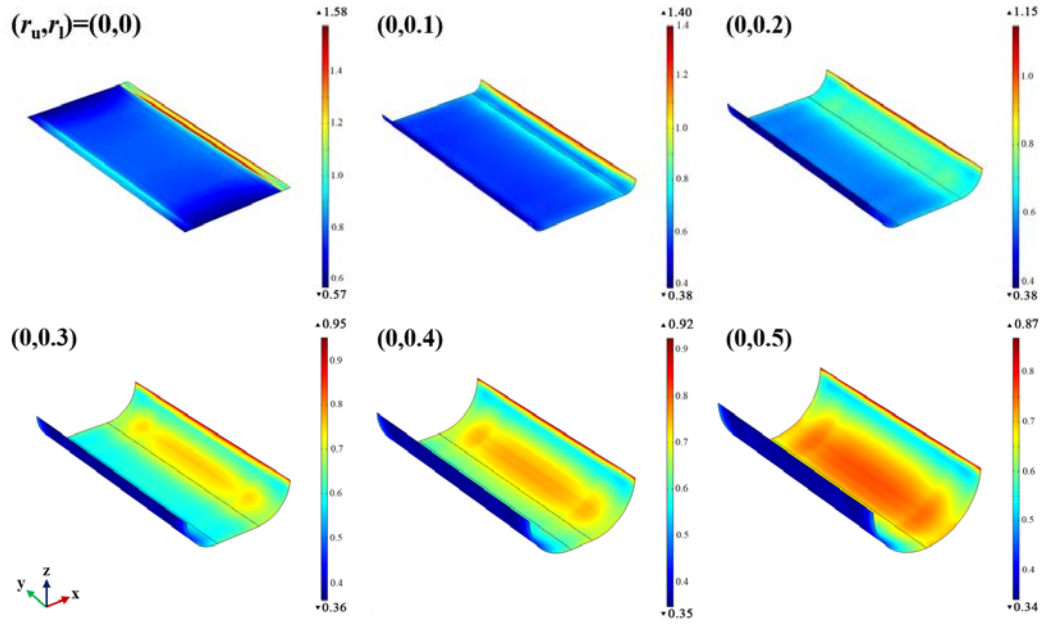
**Figure 7** The maximum von Mises stress in (a) TE module; (b) electrode; and (c) TE material for the U-type unileg thermoelectric module as a function of the lower ( $r_l$ ) and upper ( $r_u$ ) radii, under load matching conditions.



**Figure 8** The distribution of von Mises stress for the U-type unileg module at (a)  $(r_u, r_l)=(0,0.3)$ ; and (b)  $(0,0.4)$ .

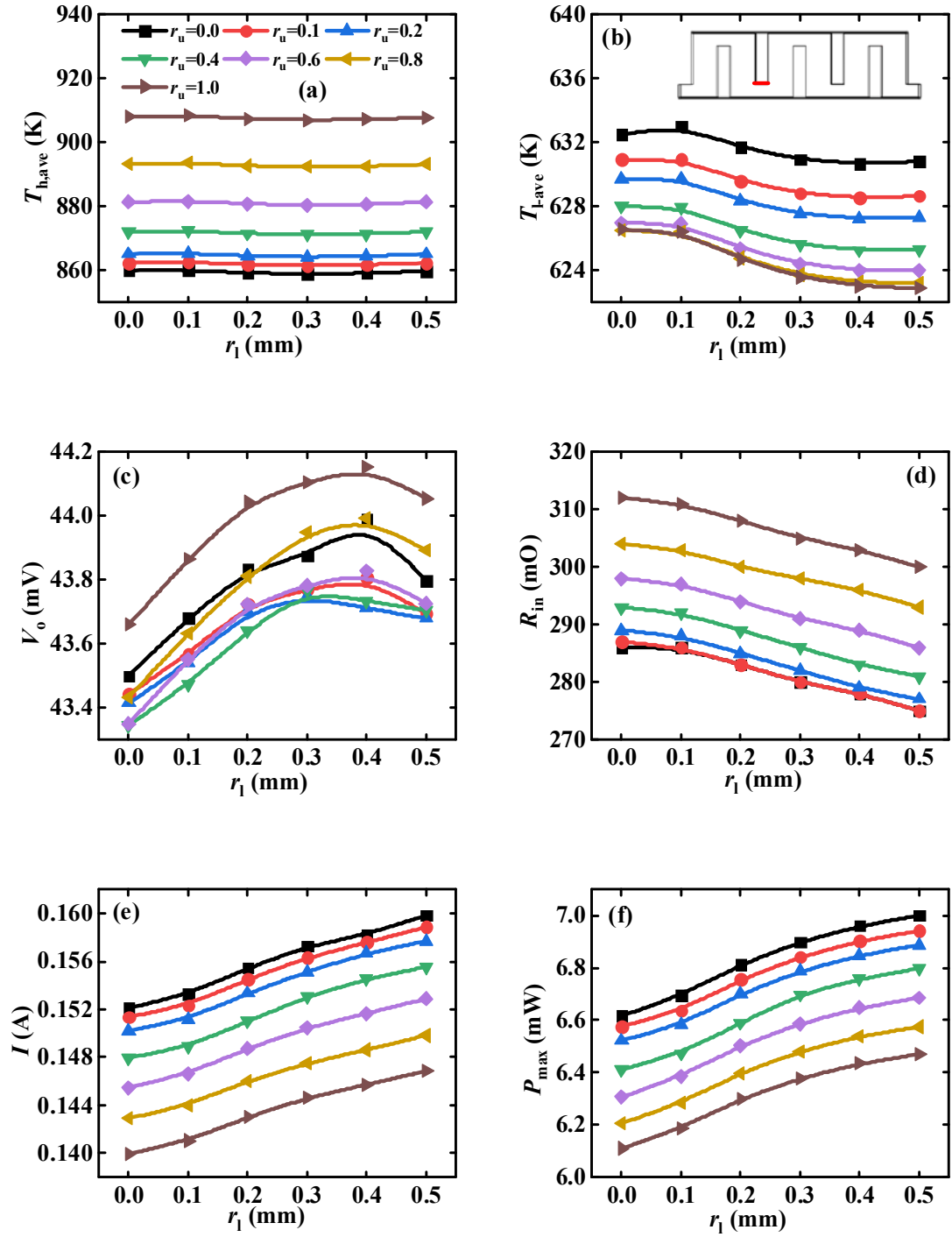


**Figure 9** The variation of von Mises stress for the electrode/ $\text{Al}_2\text{O}_3$  interface along x-axis at different upper radii.



**Figure 10** The distribution of von Mises stress for the first lower corner at different lower radii.

**Figure 11** displays the  $T_{h,ave}$ , average temperature of the first lower corner ( $T_{l,ave}$ ),  $V_o$ ,  $R_{in}$ ,  $I$ , and  $P_{max}$  of the U-type unileg module as a function of  $r_u$ , and  $r_l$ , under load matching condition, while  $Q_h$  is 6W and  $T_c$  is 300K. The  $r_u$  changes in the range of 0 to 1mm, meanwhile, the  $r_l$  varies from 0 to 0.5mm. In **Figure 11(a)**, it can be observed that  $T_{h,ave}$  monotonously enhances when  $r_u$  is larger due to the increased  $R_{th}$ , and fluctuates by about 1K with the increasing of  $r_l$ . Moreover,  $T_{l,ave}$  decreases when  $r_u$  is enlarged, while it firstly decreases and then improves as  $r_l$  raises (see **Figure 11(b)**). Consequently, the difference value between  $T_{h,ave}$  and  $T_{l,ave}$  at  $r_u=1$ mm is enlarged by about 55K compared to the value at  $r_u=0$ mm, meantime, the temperature difference is firstly declined and slightly enhanced around 5K as  $r_l$  increases.



**Figure 11** (a) Hot-side; and (b) lower corner average temperatures; (c) output voltage; (d) internal resistance; (e) working current; and (f) maximum output power, as a function of fillet radius of lower corner ( $r_1$ ) at different fillet radii of upper corner ( $r_u$ ), under load matching conditions.

Based on the temperature variation, as presented in **Figure 11** (c),  $V_o$  is firstly improved and then it is reduced with the increase of  $r_1$  and the decrease of  $r_u$ ,

achieving the peak value (44.15mV) at  $r_u$  of 1mm, and  $r_l$  of 0.4mm. In addition, higher  $r_u$  and lower  $r_l$  lead to larger material resistance, thus the  $R_{in}$  enhances with increasing  $r_u$  and decreasing  $r_l$ , as shown in **Figure 11(d)**. Then, owing to the coupled effect of  $V_o$  and  $R_{in}$ ,  $I$  diminishes when  $r_u$  increases and improves when  $r_l$  increases (see **Figure 11 (e)**). Therefore, the optimal  $P_{max}$  values are achieved at lower  $r_u$  and higher  $r_l$ . As described in **Figure 11(f)**, the maximum  $P_{max}$ , 7mW, is obtained at  $r_u=0$ mm, and  $r_l=0.5$ mm, which is 6% larger than that of the module with straight corners.

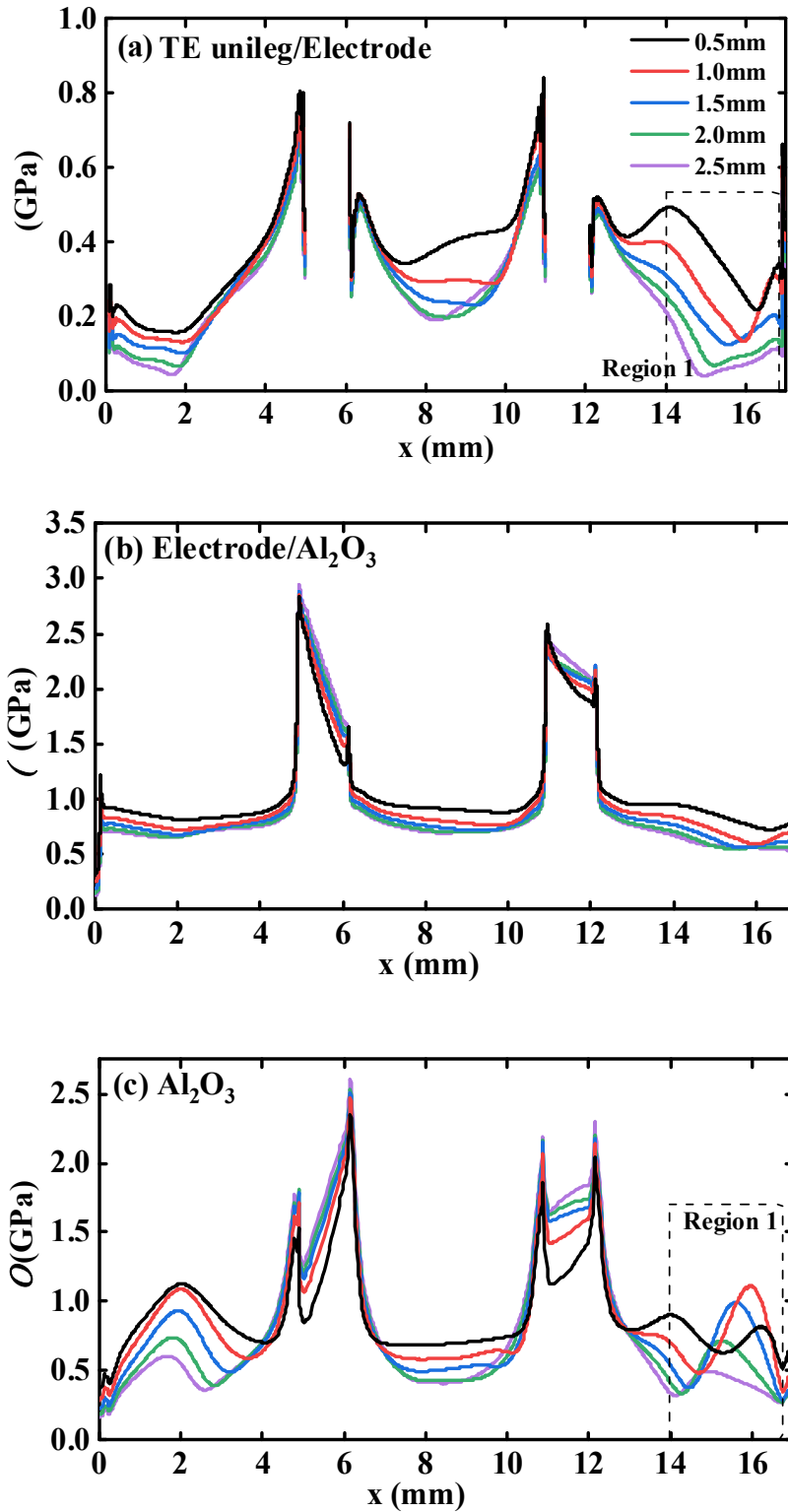
All in all, the local stress concentration and output performance are influenced by adding rounded corners. The local stress can be greatly decreased at larger  $r_u$  and  $r_l$ , while power output is enhanced at lower  $r_u$  and higher  $r_l$ . However, the structure modification results in the transfer of maximum stress position, and the lowest  $\sigma_{max,TEM}$ , obtained at  $(r_u, r_l)=(0.1, 0)$ , is only 7% lower than the obtained for the initial structure. Therefore, in order to reduce the  $\sigma_{max,TEM}$ , more geometric parameters, besides local structure, need to be systematically optimized.

## 5. The influence of geometric size

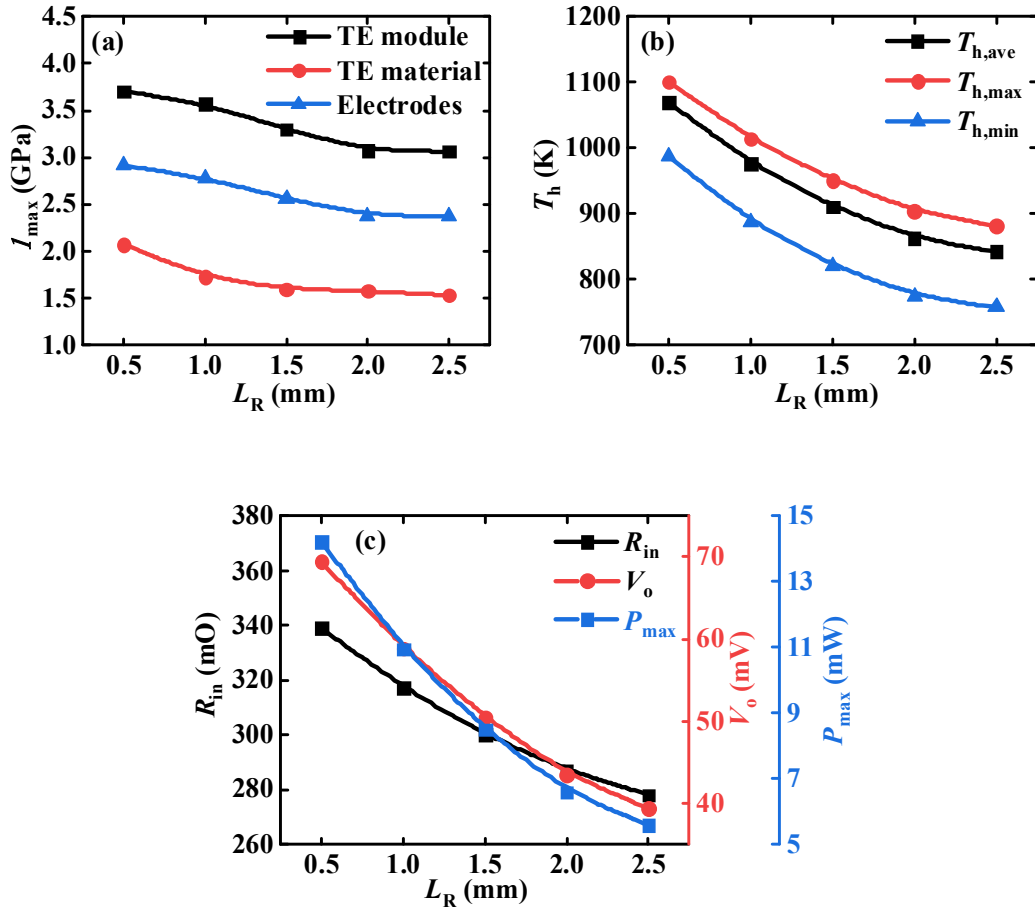
As above mentioned, geometric size is an essential factor for the location of the maximum stress and the level of thermal stress intensity. For the U-type unileg structured module, the Ag-coated right leg as conductive wire has a more complex structure, which has great influence on module output and mechanical performance. Hence, in this section, the influence of right leg length ( $L_R$ ), and Ag-coating layer thickness ( $H_{Ag}$ ), as the two most important geometric factors, on the performance and stress distribution of the optimal U-type thermoelectric module, with  $r_u=0.1$ mm and  $r_l=0$ mm, is investigated at  $Q_h=6$ W and  $T_c=300$ K.

**Figure 12** presents  $\sigma_{TEU/E}$ ,  $\sigma_{E/A}$ , and  $\sigma_A$  along the x-axis direction at different  $L_R$  under load matching conditions. The curves evolution along x-axis is roughly similar for different  $L_R$  values. The high stress occurs locally at the edge of thermoelectric unilegs, such as  $x=5, 6, 11,$  and  $12$ mm, and the stress becomes more moderate at simple structural parts. However, the stress in different regions shows various

changing trends as  $L_R$  varies. In the unileg region, the stress value is decreased when  $L_R$  is larger due to the lower temperature achieved. In the junction region, the stress is enhanced at larger  $L_R$ , as shown in **Figures 12(b)** and **12(c)**. Moreover, the local range of stress variation is also diverse due to the structural changes. Taking the region 1 in **Figure 12(a)** as an example,  $\sigma_{TEU/E}$  firstly reaches the lowest point and then increases as  $x$  increases for the different  $L_R$ . The position with the lowest point gradually shifts to smaller  $x$  with the increase of  $L_R$ , which agrees well with the modified geometric structure. Then, at  $x=16.33-16.64\text{mm}$  local position, the maximum  $\sigma_{TEU/E}$  is obtained for  $L_R=1.0\text{mm}$ , which is nor the largest or smallest  $L_R$ . As for  $\sigma_A$  in region 1, the value firstly declines, later rises and, finally, declines again.



**Figure 12** Von Mises stress of (a) thermoelectric unileg/electrode interface; (b) electrode/ $\text{Al}_2\text{O}_3$  interface; and (c)  $\text{Al}_2\text{O}_3$  surface along x-axis, with different right legs length, under load matching conditions.



**Figure 13** (a) Maximum von Mises stress of thermoelectric module, thermoelectric material, and electrodes; (b) average, maximum, and minimum hot-side temperature; (c) working current, output voltage, and maximum output power for the U-type thermoelectric unileg module with  $r_u=0.1\text{mm}$ , and  $r_l=0\text{mm}$ , as a function of right legs length, under load matching conditions.

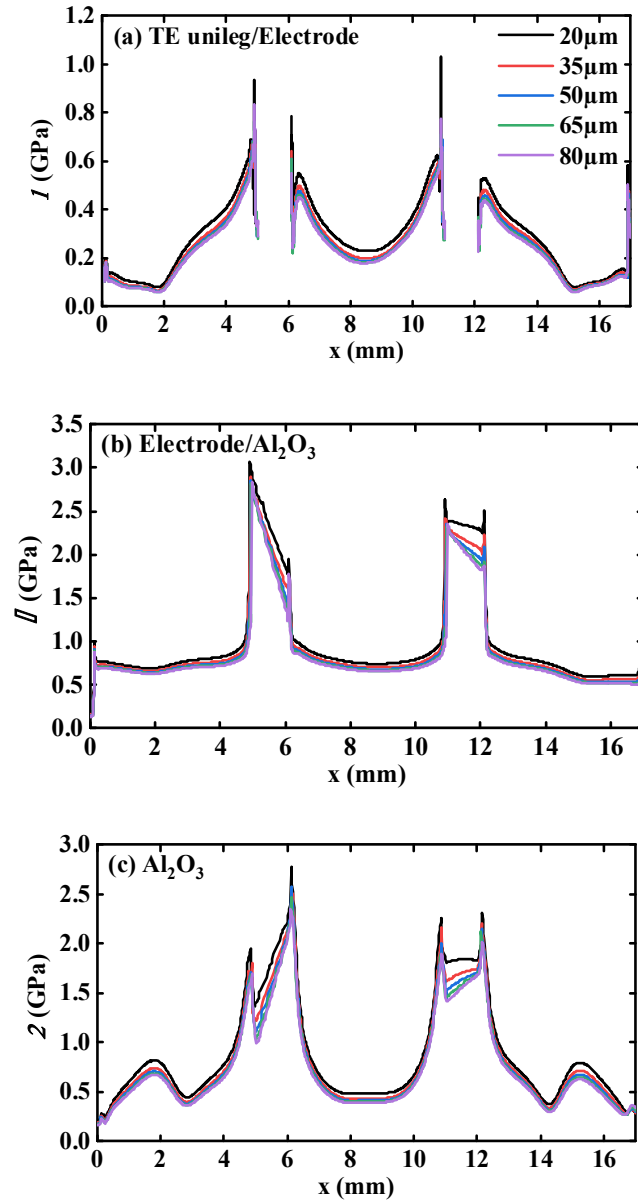
Furthermore, the extreme point is achieved at a smaller  $x$  as  $L_R$  increases. In conclusion, as plotted in **Figure 13(a)**,  $\sigma_{\max}$  of different components tends to decrease as the  $L_R$  increases from 0.5 to 2.5mm. The phenomenon can be reasoned taking into account two factors: On one hand, smaller  $L_R$  results in higher overall  $R_{th}$  per surface area and larger temperature gradient. On the other hand, rapid changes in local structure lead to stress concentration. Thus, in order to achieve higher mechanical performance, larger length of right legs should be chosen. **Figure 13(b) and 13(c)** present  $T_h$ ,  $R_{in}$ ,  $V_o$ , and  $P_{\max}$  of the optimal U-type thermoelectric module, as a function of  $L_R$ . As it can be seen in the figure, the increase of  $L_R$  leads to decreasing



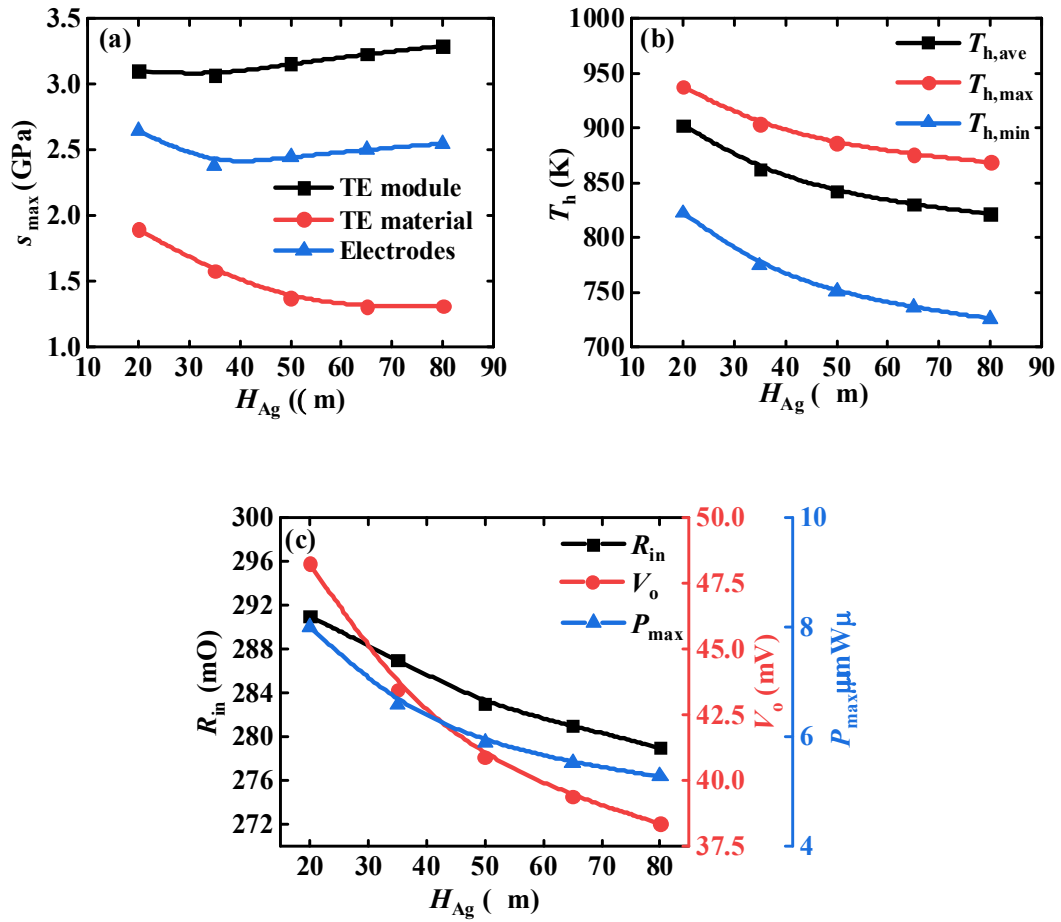
$R_{in}$  due to the enlarged cross-sectional area. The variation tendency nearly fits with the results obtained through the equation of shunt resistance,  $R_{in} \sim (1/(R_{left}^{-1} + kL_R))$ , where  $R_{left}$  is resistance of left leg and  $k$  is a coefficient depending on geometric sizes and electrical resistivity. However, the increased  $L_R$  also results in a lower  $R_{th}$ , leading to smaller  $\Delta T$  at a certain  $Q$ , as shown in **Figure 13(b)**, and, subsequently, diminished  $V_o$ . Moreover,  $V_o$  shows a more significant reduction compared to  $R_{in}$  and, consequently,  $P_{max}$  finally drops from 14.2 to 5.6mW when  $L_R$  increases from 0.5 to 2.5mm. In a word, the optimization strategies for mechanical and output performance are opposite. Greater  $L_R$  deteriorates the module performance, but it is still beneficial for mechanical properties by decreasing the peak thermal stress and dispersing the high thermal stress regions. Hence, after comprehensive consideration, the  $L_R$  of 1.5-2mm is considered as the optimal size because of the low gradient of  $\sigma_{max}$  at  $L_R > 2$ mm.

The thickness of Ag-coating layer is another important factor that affects thermal stress distribution. **Figure 14** displays  $\sigma_{TEU/E}$ ,  $\sigma_{E/A}$ , and  $\sigma_A$  along x-axis for different  $H_{Ag}$  values. For the local hot-side stress, the values are lower for larger  $H_{Ag}$  due to the induced decrease of  $T_h$ . To more comprehensively study the change rules, **Figure 15(a)** shows the relationship between  $\sigma_{max}$  of different components and  $H_{Ag}$ . In this case,  $\sigma_{max,TEM}$  is larger than  $\sigma_{max,E}$ , and  $\sigma_{max,TE}$  at any  $H_{Ag}$  value, indicating that the maximum stress always occurs on the  $Al_2O_3$  substrate. Moreover, the stress firstly decreases and then increases as  $H_{Ag}$  increases. The minimum point of  $\sigma_{max,TEM}$ , and  $\sigma_{max,E}$  is achieved for  $H_{Ag}=35\mu m$ , while the lowest  $\sigma_{max,TE}$  value is obtained for  $H_{Ag}=65\mu m$ . **Figures 15(b)** and **15(c)** depict  $T_h$ ,  $R_{in}$ ,  $V_o$ , and  $P_{max}$  of the optimal U-type thermoelectric module as a function of  $H_{Ag}$ , at  $Q_h=6W$  and  $T_c=300K$ , under load matching conditions. When  $H_{Ag}$  is increased from 20 to 80 $\mu m$ ,  $R_{th}$  is reduced and, consequently,  $T_{h,ave}$  is monotonously decreased from 902.4 to 821.3K. The decreased  $T_{h,ave}$  causes, in turn, the reduction of  $V_o$ . Then, although  $R_{in}$  shows a downwards trend as  $H_{Ag}$  increases,  $P_{max}$  drops from 8.0 to 5.3mW when  $H_{Ag}$  is enhanced from 20 to 80 $\mu m$ . In conclusion, although the stress concentration at hot end is slightly severe, a relatively thinner Ag-coating layer of about 35 $\mu m$  is beneficial for relieving the overall stress and enhancing module performance simultaneously.

In brief, greater  $L_R$  results in worse module performance but better mechanical properties. Moreover, the optimal  $P_{\max}$  and lowest  $\sigma_{\max, \text{TEM}}$  are all obtained with a relatively thin Ag-coating layer. Above results proved that different performance demands will be satisfied through the optimization of geometric size, however, the final design for any practical devices should be proposed after considering the actual performance requirements.



**Figure 14** Von Mises stress of (a) thermoelectric unileg/electrode interface; (b) electrode/ $\text{Al}_2\text{O}_3$  interface; and (c)  $\text{Al}_2\text{O}_3$  surface along x-axis for different Ag layer thickness.



**Figure 15** (a) Maximum von Mises stress of thermoelectric module, thermoelectric material, and electrodes; (b) average, maximum, and minimum hot-side temperature; and (c) working current, output voltage, and maximum output power for the U-type thermoelectric unileg module with  $r_u=0.1\text{mm}$  and  $r_l=0\text{mm}$ , as a function of Ag layer thickness, under load matching conditions.

## 6. Conclusion

A  $\text{CaMnO}_3$ -based U-type thermoelectric module combining unileg structure and pn-junction has been for the first time, in the best of our knowledge, proposed and investigated in this paper. In this structure, right leg is practically short-circuited through coating it with conductive material, acting as an electrical conductor for the unileg structure. The proposed structure avoids the module failure due to high thermal stress, produced at its hot side, between the thermoelectric materials and the electrode. Moreover, the need for soldering at the hot end of the thermoelectric unilegs is eliminated, and the module can keep working even if the hot-side electrode is broken.

The U-type unileg module can achieve  $P_{\max}$  of 6.6mW,  $\sigma_{\max}$  of 3.31GPa and  $N$  of 41686 cycles at  $Q_h=6W$  and  $T_c=300K$ . The  $\sigma_{\max}$  and  $N$  are 46% and 132% of those obtained in traditional unileg modules with the same material and module size, indicating its high thermal stability. Based on the structural modifications performed to decrease the thermal stresses, the effect of  $r_u$  and  $r_l$  has been investigated. The results showed that lower  $r_u$  and higher  $r_l$  are beneficial for improving power output. On the other hand, larger  $r_u$  and  $r_l$  are suitable to relieve the local stress concentration; however, the lowest stress of overall module has been obtained for  $r_u=0.1mm$ , and  $r_l=0mm$ . Envisaging further decreasing the stress and improving module performance, the effect of  $L_R$  and  $H_{Ag}$  has also been considered. The results showed that greater  $L_R$  deteriorates the module performance, but it is beneficial for mechanical properties by decreasing the peak stress and dispersing the high thermal stress regions. Moreover, a relatively thin Ag-coating layer (35 $\mu$ m) is suitable for obtaining high  $P_{\max}$  and relieving  $\sigma_{\max,TEM}$  simultaneously. To summarize, the U-type thermoelectric module has shown to have high thermal and mechanical stability under high temperatures and thermal cycling conditions, which would be highly beneficial for commercializing high-temperature thermoelectric devices with excellent mechanical strength and long lifespan.

### **Declaration of interests**

There are no conflicts to declare.

### **Credit authorship contribution statement**

**Xue Wang:** Conceptualization, Data curation, Investigation, Formal analysis, Visualization, Writing-Original draft preparation. **Hongchao Wang:** Supervision, Project administration, Writing-Reviewing and Editing. **Wenbing Su:** Resources. **Tingting Chen:** Software. **Chang Tan:** Validation. **M. A. Madre:** Validation. **A. Sotelo:** Conceptualization, Writing-Reviewing and Editing. **Chunlei Wang:** Supervision.

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## **Nomenclature:**

TEG: thermoelectric power generator

$T$ : temperature (K)

$\Delta T$ : temperature difference (K)

$\rho_d$ : density ( $\text{kg/m}^3$ )

$E$ : Young's modulus (GPa)

$\nu$ : Poisson's rate

CTE: coefficient of thermal expansion ( $10^{-6}/\text{K}$ )

$S$ : Seebeck coefficient (V/K)

$\rho$ : electrical resistivity ( $\Omega\text{m}$ )

$\kappa$ : thermal conductivity (W/mK)

$P_c$ : Peltier coefficient (V)

$C_p$ : heat capacity (J/kgK)

$R_{in}$ : inner resistance ( $\Omega$ )

$R_L$ : load resistance ( $\Omega$ )

$V$ : output voltage (V)

$V_{oc}$ : open-circuit voltage (V)

$I$ : working current (A)

$P$ : output power (W)

$Q_h$ : absorbed heat in the hot side (W)

$\eta$ : conversion efficiency

$\sigma$ : von Mises stress (GPa)

$N$ : number of full cycles to failure

$N_f$ : number of load reversals

$r_u$ : fillet radius of upper corners (mm)

$r_l$ : fillet radius of lower corners (mm)

$H_{Ag}$ : Ag layer thickness ( $\mu\text{m}$ )

$L_R$ : length of right legs (mm)

Subscripts:

h: hot side

c: cold side

ave: average value

max: maximum value

min: minimum value

a: actual

r: reverse

TEM: Thermoelectric module

TE: Thermoelectric material

E: Electrodes

A: upper surface of  $\text{Al}_2\text{O}_3$  substrate

TEU/E: thermoelectric unileg/electrode interface

E/A: electrode/ $\text{Al}_2\text{O}_3$  interface

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# Electronic Supplementary Information

## U-type Unileg Thermoelectric Module: A Novel Structure for High-temperature Application with Long Lifespan

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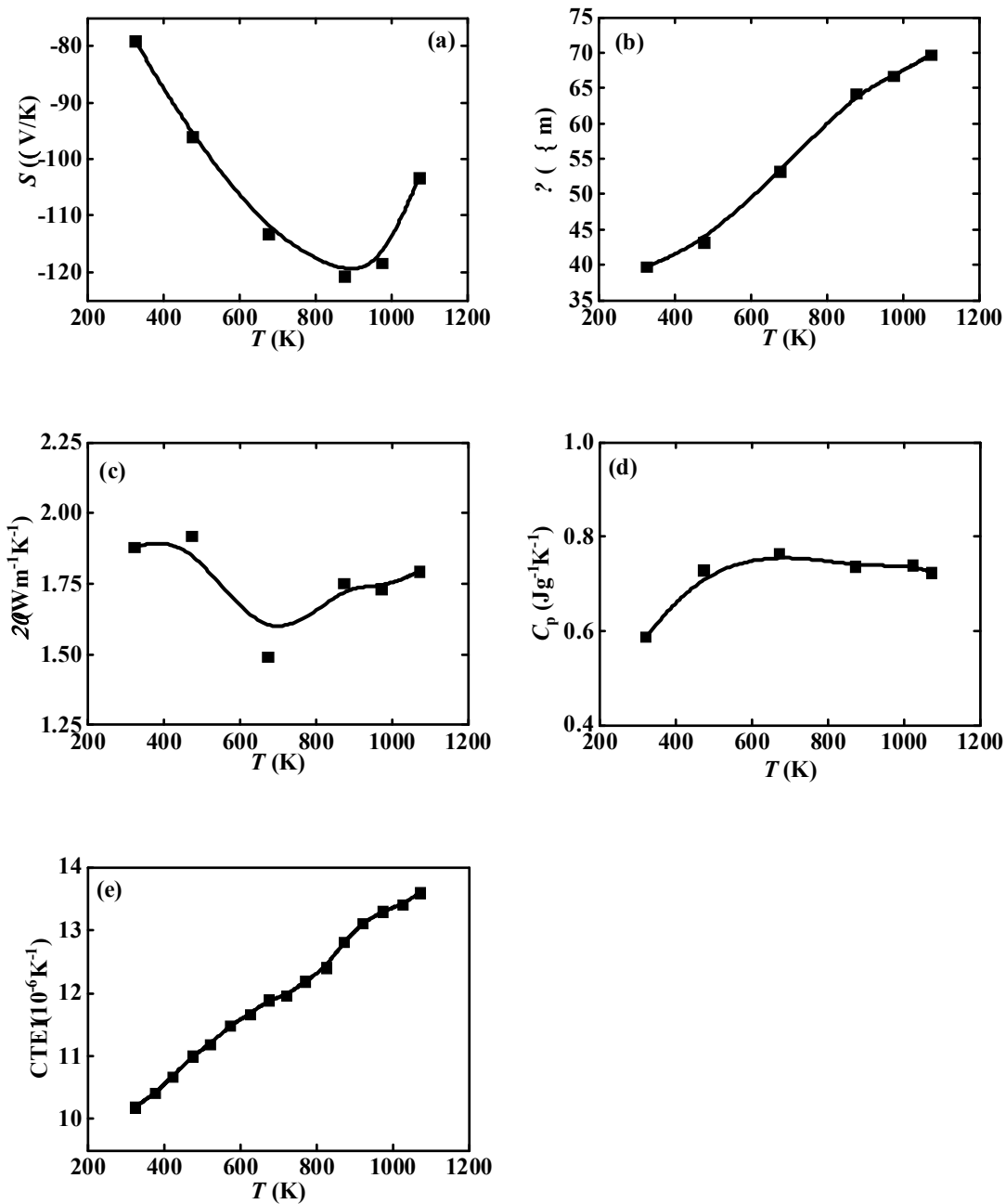
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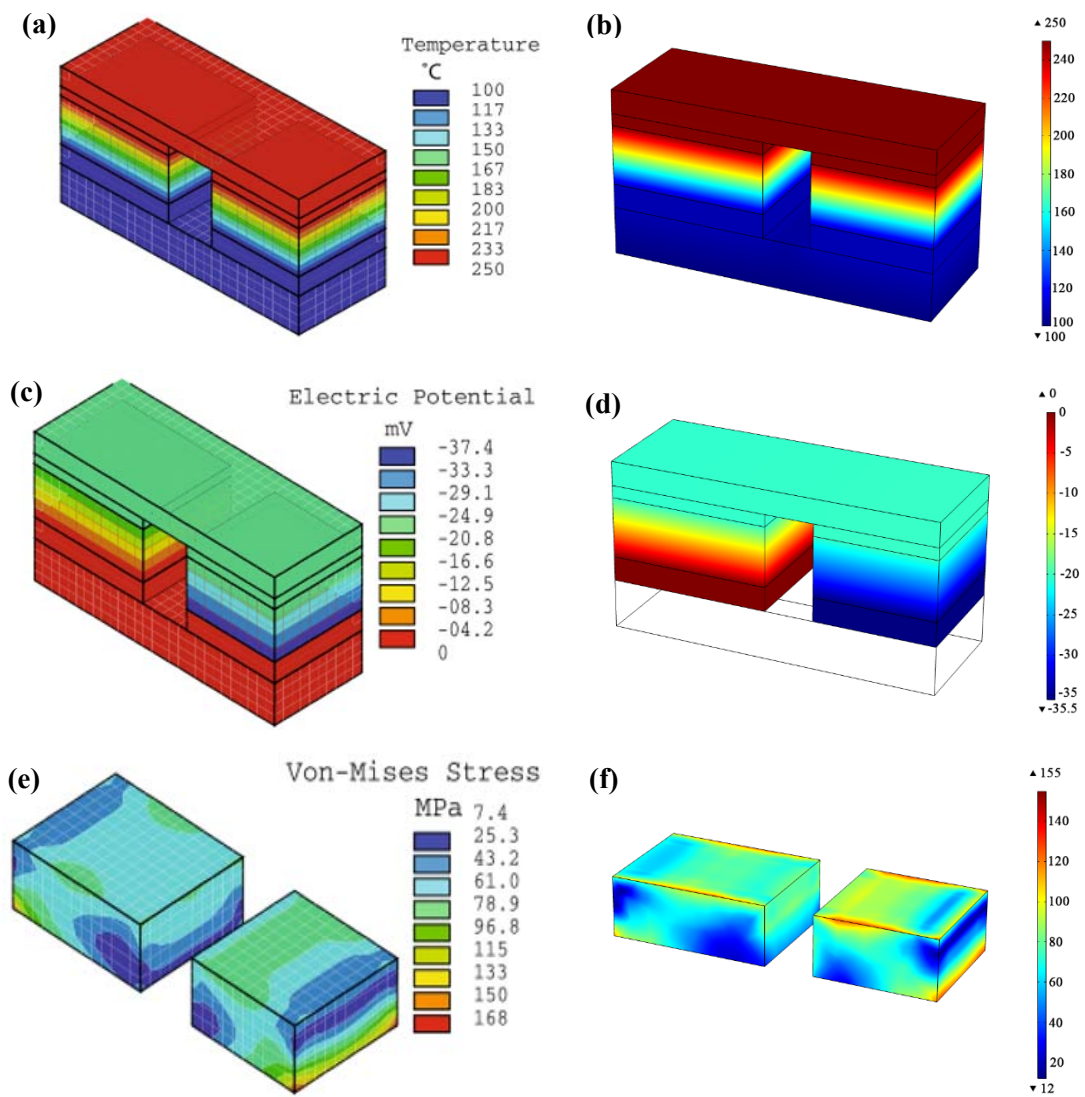
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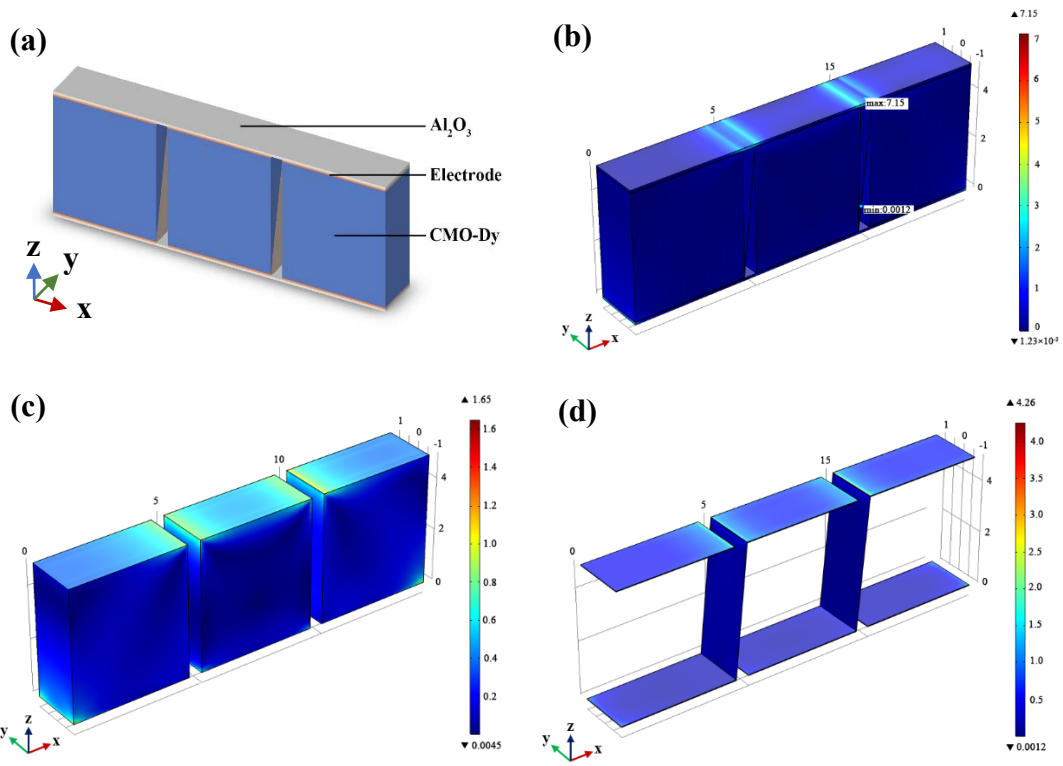
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**Figure S1** (a) Seebeck coefficient; (b) electric resistivity; (c) thermal conductivity; (d) heat capacity; and (e) thermal expansion coefficient of Dy-doped CaMnO<sub>3</sub>.



**Figure S2** Validation of present simulations with previous study. Distributions of (a) (b) temperature; (c) (d) voltage; and (e) (f) von Mises stress.



**Figure S3** (a) Schematics of traditional unileg modules. The von Mises stress distributions in (b) traditional unileg module; (c) thermoelectric leg; and (d) electrode under open-circuit conditions.

**Table S1** Properties of CMO-Dy, Ag and Al<sub>2</sub>O<sub>3</sub>.

<b>Material</b>	<b>CMO-Dy</b>	<b>Ag</b>	<b>Al<sub>2</sub>O<sub>3</sub></b>
$\rho$ (kg/m <sup>3</sup> )	$\rho(T)$	10500	3965
$E$ (GPa)	142	83	350
$\nu$	0.3	0.37	0.3
CTE (10 <sup>-6</sup> /K)	$CTE(T)$	18.9	8.8
$C_p$ (J/kgK)	$C_p(T)$	235	730
$k$ (W/mK)	$k(T)$	429	35