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We report a new family of liquid crystalline porphyrin-core dendrimers with coumarin units at the periphery of the dendrimer. These compounds have been prepared by copper-catalyzed azide-alkyne "click" cycloaddition (CuAAC). The mesomorphic properties have been investigated by polarized optical microscopy (POM), differential scanning calorimetry (DSC) and X-ray diffraction (XRD). The peripheral coumarin units play a key role in the liquid crystal behavior, contributing to the appearance of discotic nematic mesophases with hole mobility values among the highest values reported for discotic liquid crystals (of the order of $1 \text{ cm}^2 \cdot V^{-1} \cdot S^{-1}$). It has also been demonstrated that excitation of the coumarin moieties leads to energy transfer (antenna effect) to the luminescent porphyrin core. Therefore, this strategy, that involves 'click' chemistry, has proven to be a powerful and elegant synthetic tool for the preparation of optoelectronic materials based on complex dendritic architectures.

Introduction

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In the last few years the research field of organic electronics i.e. the use of conjugated molecules as active components in electronic devices- has expanded impressively and diversified into organic light-emitting diodes (OLEDs), organic field-effect transistors (OFETs) or organic photovoltaic devices (OPVs). The ideal organic semiconductor must have closely-packed molecules with few defects between molecules or domains because charge transport depends on molecular order.¹⁻⁴ The highest charge carrier mobility in organic systems has been measured in single crystals of pentacene and rubrene (20 cm²V⁻¹s⁻¹),⁵ but the preparation of single-crystalline thin films is very tedious and not applicable industrially. To overcome this limitation, liquid crystals (LCs) offer an interesting approach as they self-organize into nanostructured phases that provide properties similar to those of organic single crystals, while the dynamics is vital for the processability and the self-healing of structural defects.⁶⁻¹⁴ In particular, columnar LCs have attracted considerable attention because they represent one-dimensional charge carrier pathways, whereby the inner aromatic core is conducting and the outer alkyl periphery has an insulating character. In such phases, charge mobilities of the order of 1 cm²·V⁻¹·s⁻¹ along the direction of

the columns have been described.¹⁵⁻¹⁹ Despite this promising properties, columnar mesophases still suffer from the disadvantage that orientationally uniform domains that are large enough to be used in most devices are often hard to obtain. Nonetheless, very recently we reported a new family of LC dendrimers with charge mobilities of the same order of magnitude as those found in columnar phases, in spite of exhibiting a discotic nematic (N_D) phase.²⁰ The N_D mesophase makes possible to obtain large homogeneously oriented domains with very low density of defects due to its lower order degree in comparison to columnar mesophases.

On the other hand, LC dendrimers have been recognized as an attractive tool for the preparation of polymeric materials for optoelectronic applications.²¹⁻²⁴ They combine in the same molecule the structural and synthetic versatility of dendrimers with the anisotropic properties of the LC state. The topology of these molecules bearing several external reactive groups allows the introduction of different active moieties that self-assemble into a LC arrangement with a supramolecular order that facilitates charge transport.²⁵⁻²⁹ Although the preparation of dendrimers has been improved over the past decade, their synthesis requires extremely efficient reactions. "Click" chemistry provides efficient routes for the synthesis of dendritic structures because it requires mild reaction conditions, tolerates a wide range of functional groups, and leads to high yields.³⁰⁻³⁴

The aim of the work reported here is to exploit the 'click' chemistry as an effective and elegant approach for the preparation of LC dendrimers with potential applications in organic optoelectronic devices. Herein, we describe a new approach which allows us to connect four azido-functionalized dendrons with an alkyne functionalized porphyrin core. This

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strategy gives high yields and resulted in products that can be easily purified by non-chromatographic procedures.

Following this synthetic approach, three porphyrin-core dendrimers have been synthetized and their chemical structure are shown in **Figure 1**. Two of them are the first- and second-generation dendrimers bearing coumarin moieties (**ZnP-d₁Cou** and **ZnP-d₂Cou**). The third porphyrin-core

dendrimer was prepared to evaluate the role of countrarial peripheral units in these supermolecular by steams of the system of



Figure 1. Chemical structure of the porphyrin-core dendrimers prepared by 'click' chemistry.

Results and discussion

Synthesis and structural characterization

The synthesis of the azido functionalized dendrons $(N_3-d_1Cou, N_3-d_2Cou, and N_3-d_1C12)$ and the alkyne functionalized porphyrin core (ZnP-YNE) was described in the Supporting Information.

A convergent synthetic approach was used for the preparation of the porphyrin-core dendrimers (**Scheme 1**). In the final synthetic step, the azido functionalized dendrons and the alkyne functionalized porphyrin were coupled by coppercatalyzed azide-alkyne cycloaddition (CuAAC) using THF/water as solvent and CuSO₄·5H₂O/sodium ascorbate as the catalytic system. An excess of the azide dendron (1.5 equivalents per alkyne group) was employed to ensure the completeness of the reaction and was removed by using a previously synthesized alkyne-functionalized Wang resin. The efficiency of the CuAAC coupling was determined by FTIR, NMR spectroscopy (**Figure 2a** and **Figures S1-S10**), MALDI-TOF MS (**Figures S11-S13**), and size exclusion chromatography (SEC). All characterization methods indicated that the target porphyrincore dendrimers were monodisperse.



Scheme 1. Convergent synthetic approach used for the preparation of the porphyrin-core dendrimers.

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Figure 2. (a) ¹H NMR spectra (400 MHz, CD₂Cl₂, 298K), (b) FTIR spectra, (c) MALDI mass spectrum, (d) SEC traces of ZnP-YNE (red), ZnP-d₁Cou (black) and N₃-d₁Cou (blue).

Evidence for the formation of the dendrimers was collected from the ¹H NMR spectra (**Figure 2a**), where new peaks corresponding to the formed triazol ring appeared at around 7.95 ppm (H_v) and the ethynyl proton signal of the alkyne functionalized porphyrin core disappeared upon CuAAC coupling. Furthermore, there were changes in the ¹H shift of the methylenic protons linked to the new formed triazol (H_u and H_T).

FTIR spectra also confirmed the coupling, the bands at around 3280 (\equiv C-H), 2120 (C \equiv C) and 2110 (N₃) disappeared after the CuAAC reaction (**Figure 2b**). Further confirmation of the coupling was provided by SEC analysis and MALDI-TOF mass spectroscopy. The MALDI-TOF mass spectra showed the expected peak and no other signals corresponding to dendrimers with a partial functionalization were detected (**Figure 2c**). The SEC curves showed monomodal molar mass distributions and residual precursors were not detected (**Figure 2d**). As can be observed, CuAAC coupling of the precursors blocks gave rise to a shift of the molar mass distribution towards lower retention times that indicates the porphyrin-core dendrimers' formation.

Thermal stability and liquid crystalline properties

The thermal stability of all the compounds was studied by TGA. All of the samples showed good thermal stability with the 2% weight loss temperature ($T_{2\%}$) more than 150 °C above the

clearing point (**Table 1**). Thermal transitions and mesomorphic properties were studied by POM, DSC and XRD, and the results are listed in **Table 1**. Three cycles were carried out in DSC experiments and data were taken from the second cycle. In some cases, temperatures were taken from POM observations because transition peaks were not detected in DSC curves (**Figures S14-S17**).

Table 1. Thermal Stability and Transition Temperatures.

	T _{2%} (°C) ^[a]	Phase transitions ^[b]
N₃-d₁Cou	213	g 20 Cr 61 I
N ₃ -d ₂ Cou	230	g 11 N 73 ^[c] I
N_3 -d ₁ C12	241	Cr 30 I
ZnP-d₁Cou	344	g 24 N _D 112 ^[c] I
ZnP-d₂Cou	309	g 21 N _D 98 ^[c] I
ZnP-d ₁ C12	335	g 122 Cr 151 ^[c] M 155 ^[c] I

[a] Temperature at which 2% mass lost is detected in the thermogravimetric curve. [b] DSC data of the second heating process at a rate of 10 °C/min. Temperatures are read at the maximum of the peak. g: glass, Cr: crystal, N: nematic mesophase, N_D: discotic nematic mesophase, M: unidentified mesophase, I: isotropic liquid. [c] POM data.

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Only the azido-functionalized second generation dendron (N₃ $d_2 Cou)$ exhibited LC behavior. The other azido-functionalized dendrons (N₃-d₁Cou and N₃-d₁C12) were not mesomorphic. N₃-d₂Cou showed a birefringent texture by POM at room temperature on applying mechanical stress (Figure 3a).



Figure 3. POM microphotographs for: (a) N₃-d₂Cou taken at room temperature in the second cooling process, (b) ZnPd₁Cou taken at room temperature in the second cooling, (c) ZnP-d₂Cou taken at room temperature in the second cooling process, and (d) ZnP-d1C12 at 152 °C in the second heating process.

All the porphyrin-core dendrimers exhibited mesogenic properties and the thermal behavior was different depending on the compound. In the case of the porphyrin-core dendrimers with coumarin functional units around the porphyrin core (ZnP-d₁Cou, and ZnP-d₂Cou), enantiotropic LC mesophases were observed. The DSC curves showed only a glass transition freezing the mesomorphic order at room temperature, while clearing temperatures were established from POM observations. All the compounds had a high tendency to homeotropic alignment and the mesophase was observed by POM on applying mechanical stress to the samples showing birefringent textures (Figure 3).

The assignment of the mesophase type was carried out by XRD. The XRD pattern of the coumarin-containing second generation dendron (N₃-d₂Cou) showed only diffuse scattering, suggesting the absence of long-range positional order in the LC phase (Figure S18). This kind of pattern corresponds to a

nematic mesophase which has only orientational order. This nematic behavior can be explained by taking into account that the flexibility of the ester bonds allows adopting a rod-shaped geometry that locates the two dendrons in opposite directions.35 On the other hand, functionalization of the porphyrin core with the coumarin-containing dendrons gives rise to disk-like structures.³⁶⁻³⁸ Thus, the XRD pattern recorded for ZnP-d₁Cou is characteristic of a discotic nematic (N_D) mesophase. It displayed one relatively strong reflection in the low-angle region corresponding to the presence of a certain degree of short-range columnar order within the nematic phase, as has been previously described in other N_D phases,³⁹⁻ ⁴¹ and a broad halo in the high-angle region which corresponds to the liquid-like order of the disordered alkyl chains (Figure 4a). The calculated d-spacing (26 Å) for the signal in the lowangle region represents the average inter-disk distance and approximates the diameter of the porphyrin core. The spacing from the high-angle diffraction is 4.6 Å, which mainly originates from the liquid-like correlations between the conformationally disordered alkyl chains.

On the other hand, the XRD pattern of **ZnP-d₂Cou** only showed diffuse scattering in the low-angle region, and a broad diffuse halo in the high-angle region (Figure 4b). This pattern also confirms the presence of a N_D mesophase.⁴² It is noteworthy the disappearance of the intense maximum in the low-angle region in the XRD pattern of ZnP-d₂Cou (Figure 4c). The greater number of alkyl chains in ZnP-d2Cou could be responsible for these differences of XRD, since they may lateral disk-disk interactions hinder the between metalloporphyrin units, and consequently, prevent the formation of columnar alterations within the N_D phase.

With **ZnP-d₁C12**, the DSC curve showed an exothermic peak at 122 ºC (cold crystallization) and an endothermic peak corresponding both to the crystal-to-mesophase transition and the transition from mesophase to isotropic liquid. These two transitions were observed by POM and detected as only one endothermic peak in the DSC which corresponds to the sum of the two transition peaks, and thus the nature of the mesophase could not be identified by XRD. These results confirm the key role of the coumarin terminal units in the mesogenic behavior of these materials leading to the apparition of the mesophase at room temperature and increasing significantly the mesophase range. Both features are fundamental parameters for their practical applications in organic electronics.



Figure 4. (a) Room temperature XRD pattern of ZnP-d₁Cou, (b) room temperature XRD pattern of ZnP-d₂Cou, (c) 1D profiles of the XRD patterns.

 Table 2. Photophysical data in DCM solutions and in thin film.

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		λ _{abs} (nm) <i>Coumarin</i>	λ _{abs} (nm) Soret band	λ _{abs} (nm) <i>Q-bands</i>	λ _{em} (nm) <i>Coumarin</i>	λ _{em} (nm) <i>Q (0–1)</i>	λ _{em} (nm) ^{DC} <i>Q (0–2)</i>	DI: 10.103 ${\cal O}_{\sf F}^{[\sf a]}$	9/С8ТС06142 Ф _{ЕТ}
ZnP-YNE	DCM	-	426	558, 599	-	600	644	0.09	_
	film	-	426	560, 602	-	606	651	-	-
N ₃ -d ₁ Cou	DCM	324	-	-	388	-	-	-	-
	film	314	-	-	411	-	-	-	-
N ₃ -d ₂ Cou	DCM	323	-	-	388	-	-	-	-
	film	314	-	-	411	-	-	-	-
ZnP-d₁Cou	DCM	322	426	552, 595	383	603	643	0.12	16%
	film	322	434	562, 603	_[b]	605	649	-	-
ZnP-d₂Cou	DCM	321	427	552, 594	383	601	642	0.11	23%
	film	322	436	563, 604	_[b]	605	648	-	-
ZnP-d ₁ C12	DCM	-	417	540	-	600	644	0.14	-
	film	_	474	544	_	611	650	_	_

[a] The \mathcal{O}_{F} values were calculated from DCM solutions with tetraphenylporphyrin (0.11 in benzene) as standard, excitation at 420 nm. [b] Not detectable.

Absorption and emission properties.

The UV-Vis absorption and fluorescence spectra of porphyrincore dendrimers and precursors were collected on dilute solutions (10^{-5} to 10^{-7} M) in DCM, and in thin films prepared by casting DCM solutions onto quartz substrates. Relevant data are presented in **Table 2**.

The two coumarin-containing dendritic precursors, N_3 -d₁Cou and N_3 -d₂Cou, showed an absorption band in DCM solution at 324 and 323 nm respectively, and an emission band at 388 nm. Interestingly, in thin film the absorption bands showed a hypochromic shift of about 10 nm, while the emission band showed a significant bathochromic shift of 23 nm probably due to the aggregation effect. The porphyrin precursor **ZnP-YNE** showed in the UV-visible spectrum a Soret band at 426 nm, and two Q bands at 558 and 599 nm. Excitation of the Soret band resulted in the apparition of two emission bands at 606 (Q(0-1)) and 651 nm (Q(0-2)).

On the other hand, the UV-Vis absorption spectra of the three porphyrin-core dendrimers in DCM solution are, within experimental error, a combination of the spectra of the alkyne functionalized porphyrin and the corresponding azido functionalized dendron (**Figure 5a**). The absorption bands corresponding to the coumarin units showed similar values both in solution and in thin film, whereas the absorption bands corresponding to the porphyrin core in thin film exhibited a bathochromic shift of about 10 nm in relation to those in solution, due to the aggregation effect.

The fluorescence emission spectra of the three porphyrin-core dendrimers (λ_{exc} = 425 nm) consisted of two bands with a Q(0-1)-band at around 600 nm and a Q(0-2)-band at around 645 nm (**Figure 5a**). The fluorescence quantum yields were also measured with tetraphenylporphyrin as standard (\mathcal{P}_{F} = 0.11 in benzene), obtaining low quantum yields. The fluorescence spectra were also recorded in thin films. Compared to the data from DCM solutions, the emission Q(0-1) and Q(0-2) peaks are red-shifted ca. 5-10 nm.



Figure 5. a) UV-Vis absorption spectra in DCM solution (black line) and emission spectra (λ_{exc} = 425 nm) in DCM solution (blue line) and in thin film (red line) of **ZnP-d_1Cou. b)** UV-Vis absorption spectra in DCM solution (black line) and emission spectra (λ_{exc} = 320 nm) in DCM solution (blue line) and in thin film (red line) of **ZnP-d_1Cou**.

Nonetheless, when peripheral coumarin units of the dendrimers were excited at 320 nm, significant differences in

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the fluorescence emission spectra were observed between the measurements in DCM solution and in thin film. In the first case, the emission band at 388 nm (see blue line in Figure 5b) slightly overlap the Soret absorption band of the porphyrin, and thus the porphyrin core was excited. As a consequence, emission from the porphyrin core was also observed, indicating that intramolecular energy transfer occurs from coumarins donors to the porphyrin acceptor (antenna effect). The energy transfer efficiencies (Φ_{ET}) were calculated by comparing the donor emission in the presence of the acceptor relative to that in the absence of the acceptor, 43-45 and are because of the low degree of overlap between the emission spectrum of the coumarin and the absorption spectrum of the porphyrin. However, it is noteworthy that the Φ_{FT} value of the second generation dendrimer (**ZnP-d₂Cou**, Φ_{ET} = 23%) is higher than that of the first generation dendrimer (**ZnP-d₁Cou**, Φ_{ET} = 16%). This is probably because the higher number of peripheral coumarin units increase the effectivity of the lightharvesting.

The fluorescence spectra (λ_{exc} = 320 nm) were also recorded in thin film and compared to the data from DCM solutions; the coumarin emission band was totally suppressed and only emission from the porphyrin core was observed (see red line in **Figure 5b**). Therefore, the energy transfer from coumarin units to the porphyrin core is more efficient in the thin film than in DCM solution. This behavior can be attributed both to a higher number of coumarin donors around porphyrin acceptors and to the anisotropic nature of the films.

Electrochemical properties and hole mobilities.

Cyclic voltammetry experiments (CV) were performed to study the electrochemical properties of the porphyrin-core dendrimers. **ZnP-d1Cou**, **ZnP-d2Cou**, and **ZnP-d1C12** exhibited similar cyclic voltammograms with reduction and oxidation processes. The HOMO and LUMO energy levels, referred to the vacuum level were estimated from the oxidation and reduction waves (**Table 3**).

Table 3.	Electrochemical	narameters	and hole	mohilities
Table J.	LICCUOUICIIICAI	parameters	and noic	mobilities.

	E _{HOMO} (eV) ^[a]	E _{LUMO} (eV) ^[a]	μ _{hole} (cm ² ·V ⁻¹ ·s ⁻¹)
ZnP-d1Cou	-4.92	-1.33	0.15±0.05
ZnP-d₂Cou	-4.94	-1.35	0.9±0.2 ^[b]
ZnP-d ₁ C12	-4.92	-1.56	(2±1)×10 ⁻⁹

[a] $E_{HOMO} = -(E_{ox} - E_{1/2, FOC} + 4.8) \text{ eV}$. $E_{LUMO} = -(E_{red} - E_{1/2, FOC} + 4.8) \text{ eV}$. [b] Lower estimation of the charge carrier mobility (measured in areas with the lowest currents).

The charge carrier mobility measurements were performed by using the space charge-limited current (SCLC) technique. The SCLC method was chosen by considering the HOMO energy value of the materials, which matches the work function of gold electrodes ($W_{Au} \approx -5.1 \text{ eV}$) and thus ensures the formation of an ohmic contact. As a counter-electrode, indium tin oxide (ITO) was used because its work function ($W_{ITO} \approx -4.6 \text{ eV}$) is

significantly lower than the LUMO energy values and because, being transparent, it simplifies the DobserVationTofforthe macroscopic orientation by POM. All the cells, with a thickness in the 5-10 μm range, were filled by capillarity by heating the material above the temperature of the transition to the isotropic liquid. After sample filling, cells were cooled to room temperature. POM observations of cells of **ZnP-d1Cou**, and **ZnP-d2Cou** showed a strong tendency for homeotropic alignment. That is, without any surface treatment, the samples were perfectly oriented with the director (average direction the short molecular axes) perpendicular to the substrate. However, cells filled with **ZnP-d1C12**, which does not display liquid crystal properties at room temperature, showed large crystalline domains of various orientations.

The current-voltage curves typically exhibited a linear region at low applied voltages, where the behavior is ohmic. However, at higher voltages, where the current becomes space-charge-limited, a quadratic dependence was observed. In this regime, neglecting the effect of traps and the voltage dependence of mobility, the current dependence on applied voltage is described by the Mott-Gurney equation:⁴²

$$J = \frac{9}{8}\varepsilon_0\varepsilon_r\mu\frac{V^2}{d^3}$$

where J is the measured current density, μ is the charge mobility, ε_0 is the free-space permittivity, ε_r is the dielectric constant of the material, V is the applied voltage, and d is the thickness of the device. As the relative dielectric constant and the sample thickness can be easily measured, by using this equation charge mobilities can be extracted from J-V curves. The resulting hole mobilities values (μ_{hole}) are summarized in **Table 3**.

Figure 6 shows the typical *J-V* curves obtained for **ZnP-d₁Cou** and **ZnP-d₁C12** and the resulting hole mobilities were 0.15±0.05 and (2±1)×10⁻⁹ cm²·V⁻¹·s⁻¹, respectively. The low hole mobility value of **ZnP-d₁C12** can be explained by taking into account that this compound does not display liquid crystal properties at room temperature and thus all the measurements were carried out in macroscopically disordered crystalline phases. In the case of **ZnP-d₁Cou**, the introduction of coumarin units around the porphyrin core contributes to the appearance of the discotic nematic mesophase, leading to a remarkable increase of μ_{hole} by 8 orders of magnitude.



Figure 6. Typical *J*-*V* curves for: (a) $ZnP-d_1C12$ and (b) $ZnP-d_1Cou$ (circles represent experimental data and the two continuous lines represent ideal linear and quadratic dependence of *J* on *V*).

On the other hand, very high currents were measured for ZnPd₂Cou. However, the hole mobility values reported in Table 3 were obtained only in those areas of the samples with the lowest currents. Within the areas exhibiting higher currents, an ohmic regime was observed at all applied voltages, probably because the injected charges never reach a density high enough for the set-up of the space-charge regime. Films of ZnP-d₂Cou, with a thickness of the order of 100 nm, between ITO and Au electrodes were also prepared by spincoating followed by electrode evaporation, as described in the Supporting Information. However, also in this case only ohmic regimes and high currents were observed. This means that the charge mobility that was possible to measure via the SCLC technique is a lower estimation of the hole mobility in these materials and thus the real μ_{hole} is probably much higher.

The results obtained in these dendrimers reveal the key role played by the coumarin units in the hole mobility properties, either directly through an involvement in the hopping process, or indirectly through its promesogenic activity. Nevertheless, a definitive explanation for the different hole mobility values measured in ZnP-d₂Cou and ZnP-d₁Cou cannot be deduced with the reported data. However, a hint might be derived from the XRD patterns that, as illustrated above, suggest the presence of a certain degree of short-range columnar order only in **ZnP-d₁Cou** which results in a less homogeneous mesophase compared to ZnP-d₂Cou. Bearing in mind their spontaneous homeotropic alignment (i.e. a favorable macroscopic orientation between electrodes), a higher homogeneity within the mesophase of ZnP-d₂Cou might favor a more uniform alignment over large areas of the samples, leading to an increase of hole mobility. In addition, the higher number of coumarin units in **ZnP-d₂Cou** could also play an enhancing role in the increase of the hole mobility values.

Conclusions

Herein, we have developed a convenient and highly versatile route involving 'click' chemistry to prepare novel liquid crystalline dendrimers based on a porphyrin core and coumarin units at the periphery of the dendrimer. In these compounds we have demonstrated that excitation of coumarin donors results in energy transfer to the porphyrin central core. This antenna effect is more efficient in thin film than in solution.

The peripheral coumarin units play a key role in the liquid crystal behavior, contributing to the appearance of discotic nematic mesophases with high hole mobility values (of the order of $1 \text{ cm}^{2} \cdot V^{-1} \cdot s^{-1}$). To confirm this point a dendrimer analogue (i.e. without coumarin peripheral units) was prepared, and a decrease of 8 orders of magnitude in hole mobility was observed, stressing our hypothesis about the decisive influence of coumarin moieties in the final properties of the dendrimers. In general, the second generation dendrimer showed better optical (antenna effect) and electronic properties (hole mobility) than the first generation analogue. Thus, these results probe the importance of tuning such properties upon modification of dendrimer's generation.

In conclusion, we have found that combining metalloporphyrin core with luminescent Coumaria model and the same dendritic scaffold is a good strategy for the preparation of polymeric materials for optoelectronic applications. Particularly, the hole mobility values are one of the highest values reported for discotic liquid crystals to this date. Therefore, these porphyrin-core dendrimers have great potential as semiconducting soft materials with high hole mobility.

Conflicts of interest

There are no conflicts to declare.

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GRAPHICAL ABSTRACT

High hole mobility and light-harvesting in discotic nematic dendrimers prepared via 'click' chemistry

Alberto Concellón, Roberto Termine, Attilio Golemme, Pilar Romero, Mercedes Marcos, and José Luis Serrano*

We have developed a new strategy involving 'click' chemistry to to prepare porphyrin-core dendrimers that have coumarin units around the porphyrin core. These dendrimers exhibit nematic discotic phases with hole mobilities among the highest values described in liquid crystals. Moreover, we have also demonstrated that excitation of the coumarin moieties leads to energy transfer (antenna effect) to the porphyrin core.

